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Refinement of the crystal structure of UOs_2 from single-crystal X-ray diffraction data

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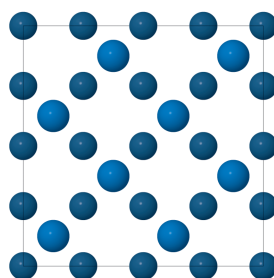
Keywords: crystal structure; intermetallics; single-crystal X-ray diffraction; redetermination; MgCu_2 structure type.

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Structural data: full structural data are available from iucrdata.iucr.org

Single crystals of uranium diosmium, UOs_2 , were obtained during an investigation of the uranium-osmium binary system by an interaction of uranium metal with osmium metal in an arc furnace and subsequent annealing. The refinement of the crystal structure of UO_2 was carried out for the first time using single-crystal X-ray diffraction data. In contrast to previous structure reports based on powder diffraction data, the U and Os atoms were refined with all U^{ij} terms of the displacement parameters considered. UOs_2 adopts the MgCu_2 structure type.

3D view



Structure description

According to the Inorganic Crystal Structure Database (ICSD; Zagorac *et al.*, 2019), the first crystal structure determination of the binary phase UOs_2 was carried out by Heal & Williams (1955) using a powder sample and a diffractometer operated with Co radiation. They reported the compound to crystallize in the space group $Fd\bar{3}m$ (No. 227) with $a = 7.5125(5) \text{ \AA}$, $V = 423.99 \text{ \AA}^3$, $Z = 8$ at 297 K belonging to the MgCu_2 structure type. Subsequent structure reports confirmed this finding with the unit-cell parameter to be in very close agreement with the one originally reported: $7.514 \text{ \AA}$ [Knapton, 1963; room-temperature data, no standard uncertainty (s.u.) given]; $7.501(4) \text{ \AA}$ (Holleck *et al.*, 1975; room-temperature data); $7.514 \text{ \AA}$ (Mentink *et al.*, 1992; room-temperature data, no s.u. given); 7.4919 (Nishioka *et al.*, 1994; room-temperature data, no s.u. given). Nishioka *et al.* (1994) also reported the existence of a polymorph of UOs_2 belonging to the MgZn_2 structure type [space group $P6_3/mmc$ (No. 194) with $a = 5.49$, $c = 8.71 \text{ \AA}$, $V = 227.35 \text{ \AA}^3$, $Z = 4$; room temperature data, no s.u. given]. All mentioned crystal-structure refinements were carried out on the basis of powder diffraction techniques with both Os and U atoms being refined isotropically. Although an anisotropic refinement of the U atom is



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equivalent to the isotropic one due to symmetry constraints, the site symmetry of the Os atom does allow non-zero off-diagonal U^{ij} terms, when refined anisotropically. Here we report our results on the crystal-structure determination of UOs_2 using single-crystal X-ray diffraction data at 100 K that allowed to use the anisotropic approximation for the displacement parameters (Fig. 1).

The unit-cell parameter of UOs_2 obtained from the single-crystal diffraction experiment (Table 1) is in good agreement with all previously reported data. The U atom resides on the special $8a$ ($\bar{4}3m$) Wyckoff position. The Os atom occupies the special $16d$ ($\bar{3}m$) Wyckoff position. The closest interatomic distances are $\text{U}\cdots\text{U}$ 3.24669 (8), $\text{Os}\cdots\text{Os}$ 2.65091 (7), and $\text{U}\cdots\text{Os}$ 3.10847 (8) Å.

Since UOs_2 is isotypic to the well known Laves phase MgCu_2 , its crystal structure will only be described very briefly. The U atoms are arranged like the atoms in the cubic diamond structure type. The Os atoms form Os_4 tetrahedra. Their virtual centres of gravity are also arranged according to the cubic diamond structure type, and the two networks interpenetrate. Overall, the coordination number of the Os atom is 12 within a slightly distorted icosahedral coordination environment. Each Os atom resides within the centre of a U_6 ring adopting a chair conformation and is additionally surrounded by six Os atoms in the shape of a trigonal antiprism. Overall, the coordination number of the U atom is 16. Each U atom is surrounded tetrahedrally by four U atoms and by twelve Os atoms in the shape of a fourfold truncated tetrahedron, which is sometimes referred to as the Friauf polyhedron (Wells, 1975).

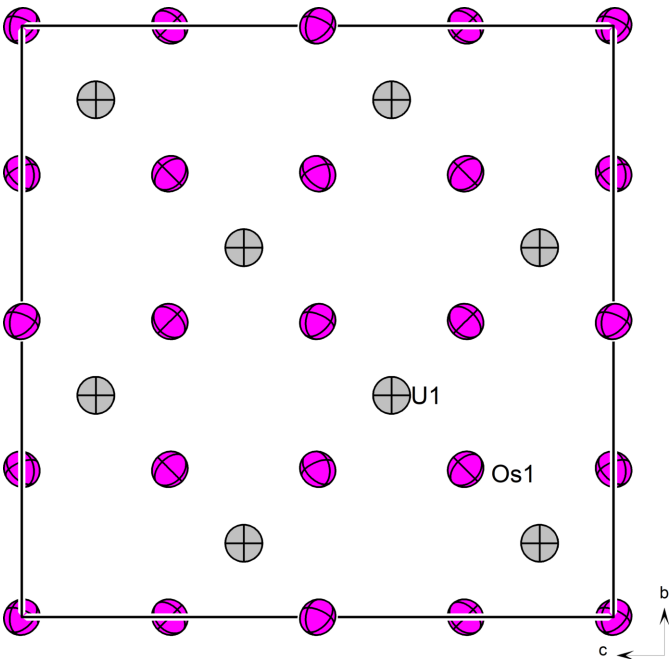


Figure 1
Crystal structure of UOs_2 in a projection along $[100]$. Displacement ellipsoids are shown at the 90% probability level.

Table 1
Experimental details.

Crystal data	UOs_2
Chemical formula	618.43
M_r	Cubic, $Fd\bar{3}m$
Crystal system, space group	100
Temperature (K)	7.4979 (2)
a (Å)	421.52 (3)
V (Å ³)	8
Z	8
Radiation type	Mo $K\alpha$
μ (mm ⁻¹)	196.43
Crystal size (mm)	$0.05 \times 0.04 \times 0.03$
Data collection	
Diffractometer	Bruker D8 VENTURE
Absorption correction	Numerical (SADABS; Krause <i>et al.</i> , 2015)
$T_{\min}$, $T_{\max}$	0.014, 0.127
No. of measured, independent and observed $[I > 2\sigma(I)]$ reflections	2221, 69, 66
R_{int}	0.083
$(\sin \theta/\lambda)_{\text{max}}$ (Å ⁻¹)	0.830
Refinement	
$R[F^2 > 2\sigma(F^2)]$, $wR(F^2)$, S	0.020, 0.047, 1.40
No. of reflections	69
No. of parameters	5
$\Delta\rho_{\text{max}}$, $\Delta\rho_{\text{min}}$ (e Å ⁻³)	1.92, -2.16

Computer programs: *APEX5* (Bruker, 2023), *SAINT* (Bruker, 2019), *SHELXT* (Sheldrick, 2015a), *SHELXL* (Sheldrick, 2015b), *DIAMOND* (Brandenburg, 2022) and *publCIF* (Westrip, 2010).

Synthesis and crystallization

Single crystals of UOs_2 were obtained by a direct reaction between stoichiometric amounts of uranium and osmium metal powders in an arc furnace. The polycrystalline reaction product was subsequently annealed in a tantalum ampule at 1073 K for 11 d and crushed in oil under a microscope to select a single crystal suitable for the diffraction study.

Refinement

Details of data collection and structure refinement are given in Table 1.

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full crystallographic data

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Uranium diosmium

Crystal data

UOs_2
 $M_r = 618.43$
 Cubic, $Fd\bar{3}m$
 $a = 7.4979$ (2) Å
 $V = 421.52$ (3) Å³
 $Z = 8$
 $F(000) = 1952$
 $D_x = 19.490$ Mg m⁻³

Mo $K\alpha$ radiation, $\lambda = 0.71073$ Å
 Cell parameters from 2037 reflections
 $\theta = 4.7\text{--}36.2^\circ$
 $\mu = 196.43$ mm⁻¹
 $T = 100$ K
 Block, metallic grey
 $0.05 \times 0.04 \times 0.03$ mm

Data collection

Bruker D8 VENTURE
 diffractometer
 Radiation source: microfocus X-ray tube
 ω and ϕ scans
 Absorption correction: numerical
 (*SADABS*; Krause *et al.*, 2015)
 $T_{\min} = 0.014$, $T_{\max} = 0.127$
 2221 measured reflections

69 independent reflections
 66 reflections with $I > 2\sigma(I)$
 $R_{\text{int}} = 0.083$
 $\theta_{\max} = 36.2^\circ$, $\theta_{\min} = 4.7^\circ$
 $h = -11 \rightarrow 10$
 $k = -10 \rightarrow 11$
 $l = -12 \rightarrow 10$

Refinement

Refinement on F^2
 Least-squares matrix: full
 $R[F^2 > 2\sigma(F^2)] = 0.020$
 $wR(F^2) = 0.047$
 $S = 1.40$
 69 reflections
 5 parameters
 0 restraints
 Primary atom site location: dual

Secondary atom site location: difference Fourier map
 $w = 1/[\sigma^2(F_o^2) + (0.0161P)^2 + 10.8153P]$
 where $P = (F_o^2 + 2F_c^2)/3$
 $(\Delta/\sigma)_{\max} < 0.001$
 $\Delta\rho_{\max} = 1.92$ e Å⁻³
 $\Delta\rho_{\min} = -2.16$ e Å⁻³
 Extinction correction: SHELXL2019/2
 (Sheldrick 2015b),
 $F_c^* = kF_c[1 + 0.001x F_c^2 \lambda^3 / \sin(2\theta)]^{-1/4}$
 Extinction coefficient: 0.00055 (10)

Special details

Geometry. All esds (except the esd in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell esds are taken into account individually in the estimation of esds in distances, angles and torsion angles; correlations between esds in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell esds is used for estimating esds involving l.s. planes.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters ($\text{\AA}^2$)

	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{iso}}^*/U_{\text{eq}}$
U1	0.875000	0.375000	0.375000	0.0088 (3)
Os1	0.500000	0.250000	0.250000	0.0082 (3)

Atomic displacement parameters ($\text{\AA}^2$)

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
U1	0.0088 (3)	0.0088 (3)	0.0088 (3)	0.000	0.000	0.000
Os1	0.0082 (3)	0.0082 (3)	0.0082 (3)	0.00062 (12)	0.00062 (12)	−0.00062 (12)

Geometric parameters ($\text{\AA}$, $^\circ$)

U1—Os1 ⁱ	3.1085 (1)	U1—Os1 ^{xi}	3.1085 (1)
U1—Os1	3.1085 (1)	U1—U1 ^{xii}	3.2467 (1)
U1—Os1 ⁱⁱ	3.1085 (1)	U1—U1 ^{xiii}	3.2467 (1)
U1—Os1 ⁱⁱⁱ	3.1085 (1)	U1—U1 ^{xiv}	3.2467 (1)
U1—Os1 ^{iv}	3.1085 (1)	U1—U1 ^{xv}	3.2467 (1)
U1—Os1 ^v	3.1085 (1)	Os1—Os1 ^{xvi}	2.6509 (1)
U1—Os1 ^{vi}	3.1085 (1)	Os1—Os1 ^{ix}	2.6509 (1)
U1—Os1 ^{vii}	3.1085 (1)	Os1—Os1 ^{xvii}	2.6509 (1)
U1—Os1 ^{viii}	3.1085 (1)	Os1—Os1 ⁱ	2.6509 (1)
U1—Os1 ^{ix}	3.1085 (1)	Os1—Os1 ⁱⁱ	2.6509 (1)
U1—Os1 ^x	3.1085 (1)	Os1—Os1 ^{xviii}	2.6509 (1)
Os1 ⁱ —U1—Os1	50.5	Os1 ⁱⁱ —U1—U1 ^{xiv}	58.5
Os1 ⁱ —U1—Os1 ⁱⁱ	50.5	Os1 ⁱⁱⁱ —U1—U1 ^{xiv}	150.5
Os1—U1—Os1 ⁱⁱ	50.5	Os1 ^{iv} —U1—U1 ^{xiv}	58.518 (1)
Os1 ⁱ —U1—Os1 ⁱⁱⁱ	144.9	Os1 ^v —U1—U1 ^{xiv}	100.025 (1)
Os1—U1—Os1 ⁱⁱⁱ	95.2	Os1 ^{vi} —U1—U1 ^{xiv}	100.025 (1)
Os1 ⁱⁱ —U1—Os1 ⁱⁱⁱ	117.0	Os1 ^{vii} —U1—U1 ^{xiv}	58.518 (1)
Os1 ⁱ —U1—Os1 ^{iv}	95.2	Os1 ^{viii} —U1—U1 ^{xiv}	150.5
Os1—U1—Os1 ^{iv}	144.9	Os1 ^{ix} —U1—U1 ^{xiv}	150.5
Os1 ⁱⁱ —U1—Os1 ^{iv}	117.0	Os1 ^x —U1—U1 ^{xiv}	58.5
Os1 ⁱⁱⁱ —U1—Os1 ^{iv}	117.0	Os1 ^{xi} —U1—U1 ^{xiv}	58.5
Os1 ⁱ —U1—Os1 ^v	95.2	U1 ^{xii} —U1—U1 ^{xiv}	109.471 (1)
Os1—U1—Os1 ^v	117.0	U1 ^{xiii} —U1—U1 ^{xiv}	109.5
Os1 ⁱⁱ —U1—Os1 ^v	144.9	Os1 ⁱ —U1—U1 ^{xv}	58.5
Os1 ⁱⁱⁱ —U1—Os1 ^v	95.2	Os1—U1—U1 ^{xv}	58.5
Os1 ^{iv} —U1—Os1 ^v	50.5	Os1 ⁱⁱ —U1—U1 ^{xv}	100.0
Os1 ⁱ —U1—Os1 ^{vi}	144.9	Os1 ⁱⁱⁱ —U1—U1 ^{xv}	100.0
Os1—U1—Os1 ^{vi}	117.0	Os1 ^{iv} —U1—U1 ^{xv}	100.025 (1)
Os1 ⁱⁱ —U1—Os1 ^{vi}	95.2	Os1 ^v —U1—U1 ^{xv}	58.5
Os1 ⁱⁱⁱ —U1—Os1 ^{vi}	50.5	Os1 ^{vi} —U1—U1 ^{xv}	150.5
Os1 ^{iv} —U1—Os1 ^{vi}	95.2	Os1 ^{vii} —U1—U1 ^{xv}	150.5
Os1 ^v —U1—Os1 ^{vi}	117.0	Os1 ^{viii} —U1—U1 ^{xv}	58.5
Os1 ⁱ —U1—Os1 ^{vii}	117.0	Os1 ^{ix} —U1—U1 ^{xv}	58.5

Os1—U1—Os1 ^{vii}	144.9	Os1 ^x —U1—U1 ^{xv}	150.5
Os1 ⁱⁱ —U1—Os1 ^{vii}	95.2	Os1 ^{xi} —U1—U1 ^{xv}	58.5
Os1 ⁱⁱⁱ —U1—Os1 ^{vii}	95.2	U1 ^{xii} —U1—U1 ^{xv}	109.471 (1)
Os1 ^{iv} —U1—Os1 ^{vii}	50.5	U1 ^{xiii} —U1—U1 ^{xv}	109.5
Os1 ^v —U1—Os1 ^{vii}	95.2	U1 ^{xiv} —U1—U1 ^{xv}	109.5
Os1 ^{vi} —U1—Os1 ^{vii}	50.5	Os1 ^{xvi} —Os1—Os1 ^{ix}	180.0
Os1 ⁱ —U1—Os1 ^{viii}	117.0	Os1 ^{xvi} —Os1—Os1 ^{xvii}	120.0
Os1—U1—Os1 ^{viii}	95.2	Os1 ^{ix} —Os1—Os1 ^{xvii}	60.0
Os1 ⁱⁱ —U1—Os1 ^{viii}	144.9	Os1 ^{xvi} —Os1—Os1 ⁱ	60.0
Os1 ⁱⁱⁱ —U1—Os1 ^{viii}	50.5	Os1 ^{ix} —Os1—Os1 ⁱ	120.0
Os1 ^{iv} —U1—Os1 ^{viii}	95.2	Os1 ^{xvii} —Os1—Os1 ⁱ	180.0
Os1 ^v —U1—Os1 ^{viii}	50.5	Os1 ^{xvi} —Os1—Os1 ⁱⁱ	60.0
Os1 ^{vi} —U1—Os1 ^{viii}	95.2	Os1 ^{ix} —Os1—Os1 ⁱⁱ	120.0
Os1 ^{vii} —U1—Os1 ^{viii}	117.0	Os1 ^{xvii} —Os1—Os1 ⁱⁱ	120.0
Os1 ⁱ —U1—Os1 ^{ix}	95.2	Os1 ⁱ —Os1—Os1 ⁱⁱ	60.0
Os1—U1—Os1 ^{ix}	50.5	Os1 ^{xvi} —Os1—Os1 ^{xviii}	120.0
Os1 ⁱⁱ —U1—Os1 ^{ix}	95.2	Os1 ^{ix} —Os1—Os1 ^{xviii}	60.0
Os1 ⁱⁱⁱ —U1—Os1 ^{ix}	50.5	Os1 ^{xvii} —Os1—Os1 ^{xviii}	60.0
Os1 ^{iv} —U1—Os1 ^{ix}	144.9	Os1 ⁱ —Os1—Os1 ^{xviii}	120.0
Os1 ^v —U1—Os1 ^{ix}	95.2	Os1 ⁱⁱ —Os1—Os1 ^{xviii}	180.0
Os1 ^{vi} —U1—Os1 ^{ix}	95.2	Os1 ^{xvi} —Os1—U1	115.2
Os1 ^{vii} —U1—Os1 ^{ix}	144.9	Os1 ^{ix} —Os1—U1	64.8
Os1 ^{viii} —U1—Os1 ^{ix}	50.5	Os1 ^{xvii} —Os1—U1	115.2
Os1 ⁱ —U1—Os1 ^x	95.216 (1)	Os1 ⁱ —Os1—U1	64.8
Os1—U1—Os1 ^x	95.2	Os1 ⁱⁱ —Os1—U1	64.8
Os1 ⁱⁱ —U1—Os1 ^x	50.5	Os1 ^{xviii} —Os1—U1	115.2
Os1 ⁱⁱⁱ —U1—Os1 ^x	95.2	Os1 ^{xvi} —Os1—U1 ^{xix}	64.8
Os1 ^{iv} —U1—Os1 ^x	95.2	Os1 ^{ix} —Os1—U1 ^{xix}	115.2
Os1 ^v —U1—Os1 ^x	144.9	Os1 ^{xvii} —Os1—U1 ^{xix}	64.8
Os1 ^{vi} —U1—Os1 ^x	50.5	Os1 ⁱ —Os1—U1 ^{xix}	115.2
Os1 ^{vii} —U1—Os1 ^x	50.5	Os1 ⁱⁱ —Os1—U1 ^{xix}	115.2
Os1 ^{viii} —U1—Os1 ^x	144.9	Os1 ^{xviii} —Os1—U1 ^{xix}	64.8
Os1 ^{ix} —U1—Os1 ^x	117.0	U1—Os1—U1 ^{xix}	180.0
Os1 ⁱ —U1—Os1 ^{xi}	50.5	Os1 ^{xvi} —Os1—U1 ^{xv}	115.2
Os1—U1—Os1 ^{xi}	95.2	Os1 ^{ix} —Os1—U1 ^{xv}	64.8
Os1 ⁱⁱ —U1—Os1 ^{xi}	95.216 (1)	Os1 ^{xvii} —Os1—U1 ^{xv}	115.2
Os1 ⁱⁱⁱ —U1—Os1 ^{xi}	144.9	Os1 ⁱ —Os1—U1 ^{xv}	64.8
Os1 ^{iv} —U1—Os1 ^{xi}	50.5	Os1 ⁱⁱ —Os1—U1 ^{xv}	115.2
Os1 ^v —U1—Os1 ^{xi}	50.5	Os1 ^{xviii} —Os1—U1 ^{xv}	64.8
Os1 ^{vi} —U1—Os1 ^{xi}	144.9	U1—Os1—U1 ^{xv}	63.0
Os1 ^{vii} —U1—Os1 ^{xi}	95.2	U1 ^{xix} —Os1—U1 ^{xv}	117.0
Os1 ^{viii} —U1—Os1 ^{xi}	95.2	Os1 ^{xvi} —Os1—U1 ^{xx}	64.8
Os1 ^{ix} —U1—Os1 ^{xi}	117.0	Os1 ^{ix} —Os1—U1 ^{xx}	115.239 (1)
Os1 ^x —U1—Os1 ^{xi}	117.0	Os1 ^{xvii} —Os1—U1 ^{xx}	64.8
Os1 ⁱ —U1—U1 ^{xii}	150.5	Os1 ⁱ —Os1—U1 ^{xx}	115.2
Os1—U1—U1 ^{xii}	150.5	Os1 ⁱⁱ —Os1—U1 ^{xx}	64.8
Os1 ⁱⁱ —U1—U1 ^{xii}	150.5	Os1 ^{xviii} —Os1—U1 ^{xx}	115.2
Os1 ⁱⁱⁱ —U1—U1 ^{xii}	58.5	U1—Os1—U1 ^{xx}	117.0

Os1 ^{iv} —U1—U1 ^{xii}	58.5	U1 ^{xix} —Os1—U1 ^{xx}	63.0
Os1 ^v —U1—U1 ^{xii}	58.5	U1 ^{xv} —Os1—U1 ^{xx}	180.0
Os1 ^{vi} —U1—U1 ^{xii}	58.5	Os1 ^{xvi} —Os1—U1 ^{xxi}	64.8
Os1 ^{vii} —U1—U1 ^{xii}	58.5	Os1 ^{ix} —Os1—U1 ^{xxi}	115.2
Os1 ^{viii} —U1—U1 ^{xii}	58.5	Os1 ^{xvii} —Os1—U1 ^{xxi}	115.2
Os1 ^{ix} —U1—U1 ^{xii}	100.025 (1)	Os1 ⁱ —Os1—U1 ^{xxi}	64.8
Os1 ^x —U1—U1 ^{xii}	100.0	Os1 ⁱⁱ —Os1—U1 ^{xxi}	115.2
Os1 ^{xi} —U1—U1 ^{xii}	100.0	Os1 ^{xviii} —Os1—U1 ^{xxi}	64.8
Os1 ⁱ —U1—U1 ^{xiii}	100.0	U1—Os1—U1 ^{xxi}	117.0
Os1—U1—U1 ^{xiii}	58.518 (1)	U1 ^{xix} —Os1—U1 ^{xxi}	63.0
Os1 ⁱⁱ —U1—U1 ^{xiii}	58.518 (1)	U1 ^{xv} —Os1—U1 ^{xxi}	63.0
Os1 ⁱⁱⁱ —U1—U1 ^{xiii}	58.5	U1 ^{xx} —Os1—U1 ^{xxi}	117.0
Os1 ^{iv} —U1—U1 ^{xiii}	150.5	Os1 ^{xvi} —Os1—U1 ^{xiii}	115.2
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Os1 ^{vi} —U1—U1 ^{xiii}	58.5	Os1 ^{xvii} —Os1—U1 ^{xiii}	64.8
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Os1 ^x —U1—U1 ^{xiii}	58.5	U1—Os1—U1 ^{xiii}	63.0
Os1 ^{xi} —U1—U1 ^{xiii}	150.5	U1 ^{xix} —Os1—U1 ^{xiii}	117.0
U1 ^{xii} —U1—U1 ^{xiii}	109.471 (1)	U1 ^{xv} —Os1—U1 ^{xiii}	117.0
Os1 ⁱ —U1—U1 ^{xiv}	58.5	U1 ^{xx} —Os1—U1 ^{xiii}	63.0
Os1—U1—U1 ^{xiv}	100.0	U1 ^{xxi} —Os1—U1 ^{xiii}	180.0

Symmetry codes: (i) $-x+5/4, y, -z+1/4$; (ii) $-x+5/4, -y+1/4, z$; (iii) $-x+5/4, -y+3/4, z+1/2$; (iv) $-x+7/4, -y+3/4, z$; (v) $x+1/2, y+1/2, z$; (vi) $x+1/2, y, z+1/2$; (vii) $-x+7/4, y, -z+3/4$; (viii) $-x+5/4, y+1/2, -z+3/4$; (ix) $x, -y+3/4, -z+3/4$; (x) $x+1/2, -y+1/4, -z+3/4$; (xi) $x+1/2, -y+3/4, -z+1/4$; (xii) $-x+2, -y+1, -z+1$; (xiii) $-y+1, x-3/4, z+1/4$; (xiv) $-x+2, -y+1/2, -z+1/2$; (xv) $-x+3/2, -y+1, -z+1/2$; (xvi) $x, -y+1/4, -z+1/4$; (xvii) $-x+3/4, y, -z+3/4$; (xviii) $-x+3/4, -y+3/4, z$; (xix) $-x+1, -y+1/2, -z+1/2$; (xx) $x-1/2, y-1/2, z$; (xxi) $x-1/2, y, z-1/2$.