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Ti₄Fe₂C_{0.82}O_{0.18}

Huizi Liu,^a Changzeng Fan,^{a,b,*} Bin Wen^a and Lifeng Zhang^{a,c}

^aState Key Laboratory of Metastable Materials Science and Technology, Yanshan University, Qinhuangdao 066004, People's Republic of China, ^bHebei Key Lab for Optimizing Metal Product Technology and Performance, Yanshan University, Qinhuangdao 066004, People's Republic of China, and ^cSchool of Mechanical and Materials Engineering, North China University of Technology, Beijing 100144, People's Republic of China. *Correspondence e-mail: chzfan@ysu.edu.cn

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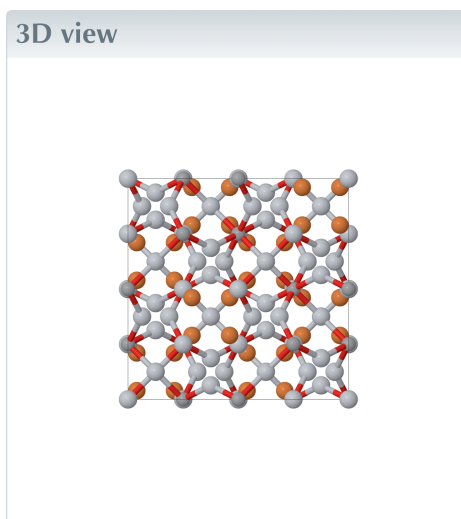
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The phase with composition Ti₄Fe₂C_{0.82}O_{0.18}, tetratitanium diiron carbide oxide, was unexpectedly synthesized by high-pressure sintering (HPS) of a stoichiometric mixture with nominal composition Ti₂Fe. The Ti₄Fe₂C_{0.82}O_{0.18} phase crystallizes in the *Fd* $\bar{3}$ *m* space group and can be considered as the Ti₂Fe structure filled with C and O atoms co-occupying the same octahedral void [occupancy ratio 0.82 (7):0.18 (7)]. The Ti₄Fe₂C_{0.82}O_{0.18} phase is isotypic with Ti₄Ni₂C and Ti₄Fe₂O_{0.407}, and is the first example where C and O atoms co-occupy the same site in filled Ti₂Fe structures.



Structure description

Intermetallic phases usually are classified by structural or chemical similarities. For example, Duwez & Taylor (1950) investigated the crystal structure of Ti₂Fe on the basis of X-ray powder data. They determined that the Ti₂Fe phase crystallizes in a face centered cubic (f.c.c.) unit cell, with cell parameter $a = 11.305 \text{ \AA}$ and with 96 atoms per unit cell. The related crystal structure of cubic Ti₄Fe₂O_{0.407} was refined on the basis of neutron powder diffraction, affording the cell parameter $a = 11.3326 (5) \text{ \AA}$ in space group *Fd* $\bar{3}$ *m* with Ti on position $48f$ and on $16d$, Fe on $32e$ and O (with a site occupation factor of 0.407) on $16c$ (Rupp & Fischer, 1988). Liu *et al.* (2024) reported the isotypic crystal structure of Ti₄Ni₂C [$a = 11.3235 (8) \text{ \AA}$] by using single-crystal X-ray diffraction (SXRD) measurements. The latter phase can be considered as a partially filled Ti₂Ni structure with the C atom occupying an octahedral void. Holleck & Thummler (1967) studied a series of carbides, nitrides and oxides in ternary systems including Nb₄Ni₂C ($a = 11.64 \text{ \AA}$) and Ta₄Ni₂C ($a = 11.61 \text{ \AA}$) phases. Although the title Ti₄Fe₂C_{0.82}O_{0.18} phase is isotypic with Ti₄Ni₂C and Ti₄Fe₂O_{0.407}, no detailed study has been performed so far with respect to a phase where C and O atoms co-occupy the same site. The carbon present in the crystal structure of Ti₄Fe₂C_{0.82}O_{0.18} most likely originated from the graphite crucible used during



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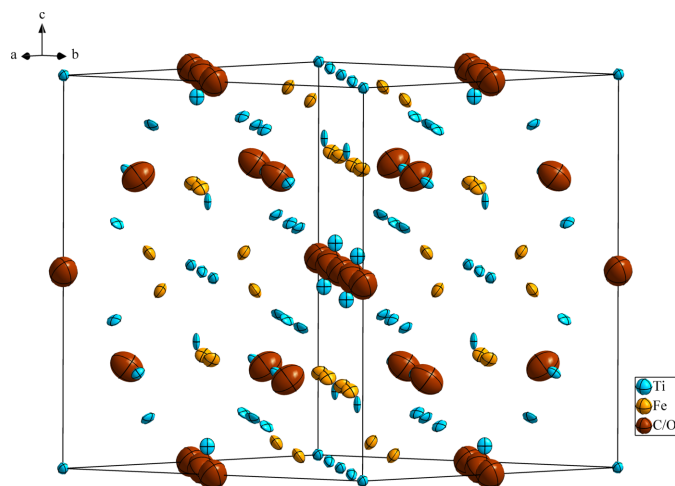


Figure 1
The crystal structure of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$, with displacement ellipsoids drawn at the 99.9% probability level.

high pressure sintering (HPS), whereas oxygen may be incorporated due to surface oxidation during sample storage and subsequent HPS.

$\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ crystallizes isotypically with $\text{Ti}_4\text{Fe}_2\text{O}_{0.407}$ and $\text{Ti}_4\text{Ni}_2\text{C}$ in a partially filled Ti_2Fe structure in space group type $Fd\bar{3}m$. Fig. 1 shows the distribution of the atoms in the unit cell of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$. The environments of the Ti1 and (C1/O1) sites are shown in Figs. 2 and 3, respectively. The Ti1 atom is situated at a position with site symmetry $\bar{3}m$ (multiplicity 16, Wyckoff letter *c*). It is surrounded by six Ti2 atoms ($2mm$, 48 *f*) and six Fe1 atoms ($3m$, 32 *e*), defining the center of an icosahedron. The (C1/O1) atoms co-occupy a position with site symmetry $\bar{3}m$ (16 *d*) and center an octahedron defined by six Ti2 atoms. The shortest Ti1 to Ti2 separation is 2.9084 (10) Å and the shortest Ti1 to Fe1 separation is 2.4650 (4) Å; the (C1/O1)–Ti2 bond length is 2.1273 (5) Å.

Synthesis and crystallization

Pure titanium powder (indicated purity 99.5%, 0.6312 g) and iron powder (indicated purity 99.9%, 0.3693 g) were evenly mixed according to the stoichiometric ratio of 2:1 and thor-

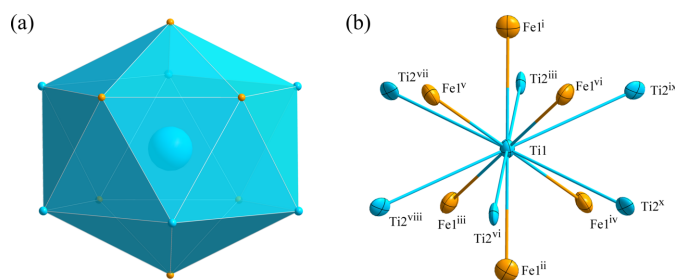


Figure 2
(a) The icosahedron formed around the Ti1 atom at the 16 *c* site; (b) the environment of the Ti1 atom with displacement ellipsoids drawn at the 99.9% probability level. [Symmetry codes: (i) $-x, y - \frac{1}{4}, z - \frac{1}{4}$; (ii) $x, -y + \frac{1}{4}, -z + \frac{1}{4}$; (iii) $-x + \frac{1}{4}, -y + \frac{1}{4}, z$; (iv) $-x + \frac{1}{4}, y, -z + \frac{1}{4}$; (v) $x - \frac{1}{4}, -y, z - \frac{1}{4}$; (vi) $x - \frac{1}{4}, y - \frac{1}{4}, -z$; (vii) $y - \frac{1}{4}, -z, x - \frac{1}{4}$; (viii) $z, -x + \frac{1}{4}, -y + \frac{1}{4}$; (ix) $-z, x - \frac{1}{4}, y - \frac{1}{4}$; (x) $-y + \frac{1}{4}, z, -x + \frac{1}{4}$.]

oughly ground in an agate mortar. The mixed powder was put into a 5 mm cemented carbide grinding mould and pressed into a tablet at about 6 MPa for 2 min to obtain a cylindrical block without deformations or cracks. Details of the high-pressure sintering experiment using a six-anvil high-temperature and high-pressure apparatus can be found elsewhere (Liu & Fan, 2018). The sample was pressurized up to 6 GPa and heated to 1473 K for 20 minutes, cooled to 1173 K, held at the temperature for 1 h, and then the furnace power was turned off to rapidly cool to room temperature. Different phases were isolated from two samples from the same batch. $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ originated from sample 1, together with TiFe. The refined chemical formula of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ from sample 1 is in accordance with the complementary EDX results (see Table S1 of the electronic supporting information, ESI). Another phase with very similar refined composition, $\text{Ti}_4\text{Fe}_2\text{C}_{0.87}\text{O}_{0.13}$, was isolated from sample 2, and its composition is also in accordance with the complementary EDX results (see Table S2 of the ESI). Different options for refinements for the two phases $\text{Ti}_4\text{Fe}_2\text{C}_{1-\delta}\text{O}_\delta$ ($\delta = 0.18$; $\delta = 0.13$) are listed in Table S3 of the ESI. The crystal structures of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ and $\text{Ti}_4\text{Fe}_2\text{C}_{0.87}\text{O}_{0.13}$ are very similar, just different in atomic proportions at the 16 *d* site, so the $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ phase was selected for the current report. Structural data for $\text{Ti}_4\text{Fe}_2\text{C}_{0.87}\text{O}_{0.13}$ can be found in Table S4 of the ESI.

Refinement

Crystal data, data collection and structure refinement details of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ are summarized in Table 1. The labeling scheme and atomic coordinates of $\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$ were adapted from $\text{Ti}_4\text{Ni}_2\text{C}$ for better comparison (Liu *et al.*, 2024). The 16 *d* site is co-occupied by C and O atoms, with site occupancies refined to 0.82 (7) for C1 and 0.18 (7) for O1, assuming full occupancy. Both atoms were refined with the same displacement parameters. The maximum and minimum residual electron densities in the final difference map are located 1.16 Å from site Fe1 and 1.67 Å from Fe1, respectively.

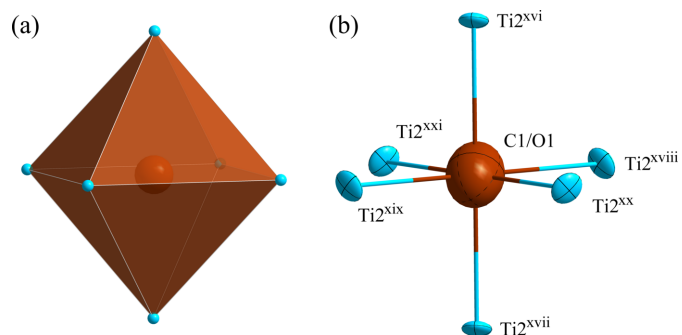


Figure 3
(a) The octahedron formed around the (C1/O1) atoms at the 16 *d* site; (b) the environment of the (C1/O1) atoms with displacement ellipsoids drawn at the 99.9% probability level. [Symmetry codes: (xvi) $-z + \frac{1}{2}, -x + 1, -y + \frac{1}{2}$; (xvii) $z + \frac{1}{2}, x, y + \frac{1}{2}$; (xviii) $x, y + \frac{1}{2}, z + \frac{1}{2}$; (xix) $-x + 1, -y + \frac{1}{2}, -z + \frac{1}{2}$; (xx) $-y + \frac{1}{2}, -z + \frac{1}{2}, -x + 1$; (xxi) $y + \frac{1}{2}, z + \frac{1}{2}, x$.]

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Table 1

Experimental details.

Crystal data	
Chemical formula	Ti ₄ Fe ₂ C _{0.82} O _{0.18}
M_r	316.02
Crystal system, space group	Cubic, $Fd\bar{3}m$
Temperature (K)	296
a (Å)	11.323 (4)
V (Å ³)	1451.6 (3)
Z	16
Radiation type	Mo $K\alpha$
μ (mm ⁻¹)	15.91
Crystal size (mm)	0.10 × 0.06 × 0.06
Data collection	
Diffractometer	Bruker D8 Venture Photon 100 CMOS
Absorption correction	Multi-scan (<i>SADABS</i> ; Krause <i>et al.</i> , 2015)
T_{\min} , T_{\max}	0.429, 0.746
No. of measured, independent and observed [$I > 2\sigma(I)$] reflections	5363, 102, 82
R_{int}	0.155
$(\sin \theta/\lambda)_{\text{max}}$ (Å ⁻¹)	0.643
Refinement	
$R[F^2 > 2\sigma(F^2)]$, $wR(F^2)$, S	0.022, 0.040, 1.14
No. of reflections	102
No. of parameters	13
$\Delta\rho_{\text{max}}$, $\Delta\rho_{\text{min}}$ (e Å ⁻³)	0.58, -0.83

Computer programs: *APEX3* and *SAINT* (Bruker, 2015), *SHELXT* (Sheldrick, 2015a), *SHELXL* (Sheldrick, 2015b), *DIAMOND* (Brandenburg & Putz, 2017) and *publCIF* (Westrip, 2010).

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full crystallographic data

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Huizi Liu, Changzeng Fan, Bin Wen and Lifeng Zhang

Tetratitanium diiron carbide oxide

Crystal data

$\text{Ti}_4\text{Fe}_2\text{C}_{0.82}\text{O}_{0.18}$
 $M_r = 316.02$
 Cubic, $Fd\bar{3}m$
 $a = 11.323$ (4) Å
 $V = 1451.6$ (3) Å³
 $Z = 16$
 $F(000) = 2342$
 $D_x = 5.784$ Mg m⁻³

Mo $K\alpha$ radiation, $\lambda = 0.71073$ Å
 Cell parameters from 1378 reflections
 $\theta = 3.1$ – 26.7°
 $\mu = 15.91$ mm⁻¹
 $T = 296$ K
 Lump, gray
 $0.10 \times 0.06 \times 0.06$ mm

Data collection

Bruker D8 Venture Photon 100 CMOS
 diffractometer
 phi and ω scans
 Absorption correction: multi-scan
 (SADABS; Krause *et al.*, 2015)
 $T_{\min} = 0.429$, $T_{\max} = 0.746$
 5363 measured reflections

102 independent reflections
 82 reflections with $I > 2\sigma(I)$
 $R_{\text{int}} = 0.155$
 $\theta_{\max} = 27.2^\circ$, $\theta_{\min} = 3.1^\circ$
 $h = -14 \rightarrow 14$
 $k = -13 \rightarrow 14$
 $l = -14 \rightarrow 14$

Refinement

Refinement on F^2
 Least-squares matrix: full
 $R[F^2 > 2\sigma(F^2)] = 0.022$
 $wR(F^2) = 0.040$
 $S = 1.14$
 102 reflections
 13 parameters

0 restraints
 $w = 1/[\sigma^2(F_o^2) + (0.0069P)^2 + 29.2966P]$
 where $P = (F_o^2 + 2F_c^2)/3$
 $(\Delta/\sigma)_{\max} < 0.001$
 $\Delta\rho_{\max} = 0.58$ e Å⁻³
 $\Delta\rho_{\min} = -0.83$ e Å⁻³

Special details

Geometry. All esds (except the esd in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell esds are taken into account individually in the estimation of esds in distances, angles and torsion angles; correlations between esds in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell esds is used for estimating esds involving l.s. planes.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters (Å²)

	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{iso}}^*/U_{\text{eq}}$	Occ. (<1)
Ti1	0.000000	0.000000	0.000000	0.0018 (5)	
Fe1	0.21037 (6)	0.21037 (6)	0.21037 (6)	0.0028 (4)	
Ti2	0.43636 (12)	0.125000	0.125000	0.0021 (4)	

C1	0.500000	0.500000	0.500000	0.012 (4)	0.82 (7)
O1	0.500000	0.500000	0.500000	0.012 (4)	0.18 (7)

Atomic displacement parameters (Å²)

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Ti1	0.0018 (5)	0.0018 (5)	0.0018 (5)	0.0004 (6)	0.0004 (6)	0.0004 (6)
Fe1	0.0028 (4)	0.0028 (4)	0.0028 (4)	−0.0007 (3)	−0.0007 (3)	−0.0007 (3)
Ti2	0.0033 (8)	0.0016 (5)	0.0016 (5)	0.000	0.000	−0.0010 (5)
C1	0.012 (4)	0.012 (4)	0.012 (4)	0.002 (4)	0.002 (4)	0.002 (4)
O1	0.012 (4)	0.012 (4)	0.012 (4)	0.002 (4)	0.002 (4)	0.002 (4)

Geometric parameters (Å, °)

Ti1—Fe1 ⁱ	2.4650 (4)	Fe1—Ti2 ^{xiii}	2.6501 (10)
Ti1—Fe1 ⁱⁱ	2.4650 (4)	Fe1—Fe1 ⁱⁱⁱ	2.734 (2)
Ti1—Fe1 ⁱⁱⁱ	2.4650 (4)	Fe1—Fe1 ⁱⁱ	2.734 (2)
Ti1—Fe1 ^{iv}	2.4650 (4)	Fe1—Fe1 ^{iv}	2.734 (2)
Ti1—Fe1 ^v	2.4650 (4)	Fe1—Ti2	2.9011 (12)
Ti1—Fe1 ^{vi}	2.4650 (4)	Fe1—Ti2 ^{xiv}	2.9011 (12)
Ti1—Ti2 ^{vii}	2.9084 (10)	Fe1—Ti2 ^{xv}	2.9011 (12)
Ti1—Ti2 ^{viii}	2.9084 (10)	Ti2—C1 ^{xvi}	2.1273 (5)
Ti1—Ti2 ^{vi}	2.9084 (10)	Ti2—C1 ^{xvii}	2.1273 (5)
Ti1—Ti2 ^{viii}	2.9084 (10)	Ti2—Ti2 ^{xviii}	2.9963 (6)
Ti1—Ti2 ^{ix}	2.9084 (10)	Ti2—Ti2 ^{xix}	2.9963 (6)
Ti1—Ti2 ^x	2.9084 (10)	Ti2—Ti2 ^{xi}	2.9963 (6)
Fe1—Ti2 ^{xi}	2.6501 (10)	Ti2—Ti2 ^{xii}	2.9963 (6)
Fe1—Ti2 ^{xii}	2.6501 (10)		
Fe1 ⁱ —Ti1—Fe1 ⁱⁱ	180.0	Ti2 ^{xi} —Fe1—Ti2 ^{xiv}	65.152 (14)
Fe1 ⁱ —Ti1—Fe1 ⁱⁱⁱ	112.64 (4)	Ti2 ^{xii} —Fe1—Ti2 ^{xiv}	124.00 (6)
Fe1 ⁱⁱ —Ti1—Fe1 ⁱⁱⁱ	67.36 (4)	Ti2 ^{xiii} —Fe1—Ti2 ^{xiv}	65.152 (14)
Fe1 ⁱ —Ti1—Fe1 ^{iv}	112.64 (4)	Fe1 ⁱⁱⁱ —Fe1—Ti2 ^{xiv}	112.83 (2)
Fe1 ⁱⁱ —Ti1—Fe1 ^{iv}	67.36 (4)	Fe1 ⁱⁱ —Fe1—Ti2 ^{xiv}	112.83 (2)
Fe1 ⁱⁱⁱ —Ti1—Fe1 ^{iv}	67.36 (4)	Fe1 ^{iv} —Fe1—Ti2 ^{xiv}	61.89 (3)
Fe1 ⁱ —Ti1—Fe1 ^v	67.36 (4)	Ti2—Fe1—Ti2 ^{xiv}	118.474 (12)
Fe1 ⁱⁱ —Ti1—Fe1 ^v	112.64 (4)	Ti1 ⁱⁱ —Fe1—Ti2 ^{xv}	65.047 (6)
Fe1 ⁱⁱⁱ —Ti1—Fe1 ^v	112.64 (4)	Ti1 ⁱⁱⁱ —Fe1—Ti2 ^{xv}	166.80 (5)
Fe1 ^{iv} —Ti1—Fe1 ^v	180.0	Ti1 ^{iv} —Fe1—Ti2 ^{xv}	65.047 (6)
Fe1 ⁱ —Ti1—Fe1 ^{vi}	67.36 (4)	Ti2 ^{xi} —Fe1—Ti2 ^{xv}	124.00 (6)
Fe1 ⁱⁱ —Ti1—Fe1 ^{vi}	112.64 (4)	Ti2 ^{xii} —Fe1—Ti2 ^{xv}	65.152 (14)
Fe1 ⁱⁱⁱ —Ti1—Fe1 ^{vi}	180.0	Ti2 ^{xiii} —Fe1—Ti2 ^{xv}	65.152 (14)
Fe1 ^{iv} —Ti1—Fe1 ^{vi}	112.64 (4)	Fe1 ⁱⁱⁱ —Fe1—Ti2 ^{xv}	61.89 (3)
Fe1 ^v —Ti1—Fe1 ^{vi}	67.36 (4)	Fe1 ⁱⁱ —Fe1—Ti2 ^{xv}	112.83 (2)
Fe1 ⁱ —Ti1—Ti2 ^{vii}	64.739 (17)	Fe1 ^{iv} —Fe1—Ti2 ^{xv}	112.83 (2)
Fe1 ⁱⁱ —Ti1—Ti2 ^{vii}	115.261 (17)	Ti2—Fe1—Ti2 ^{xv}	118.474 (12)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ^{vii}	58.40 (3)	Ti2 ^{xiv} —Fe1—Ti2 ^{xv}	118.474 (12)
Fe1 ^{iv} —Ti1—Ti2 ^{vii}	115.261 (17)	C1 ^{xvi} —Ti2—C1 ^{xvii}	140.40 (7)

Fe1 ^v —Ti1—Ti2 ^{vii}	64.739 (17)	C1 ^{xvi} —Ti2—Fe1 ^{xx}	88.009 (14)
Fe1 ^{vi} —Ti1—Ti2 ^{vii}	121.60 (3)	C1 ^{xvii} —Ti2—Fe1 ^{xx}	88.009 (14)
Fe1 ⁱ —Ti1—Ti2 ⁱⁱⁱ	58.40 (3)	C1 ^{xvi} —Ti2—Fe1 ^{xxi}	88.009 (14)
Fe1 ⁱⁱ —Ti1—Ti2 ⁱⁱⁱ	121.60 (3)	C1 ^{xvii} —Ti2—Fe1 ^{xxi}	88.009 (14)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ⁱⁱⁱ	64.739 (17)	Fe1 ^{xx} —Ti2—Fe1 ^{xxi}	168.22 (7)
Fe1 ^{iv} —Ti1—Ti2 ⁱⁱⁱ	64.739 (17)	C1 ^{xvi} —Ti2—Fe1 ⁱⁱ	137.91 (5)
Fe1 ^v —Ti1—Ti2 ⁱⁱⁱ	115.261 (17)	C1 ^{xvii} —Ti2—Fe1 ⁱⁱ	81.68 (3)
Fe1 ^{vi} —Ti1—Ti2 ⁱⁱⁱ	115.261 (17)	Fe1 ^{xx} —Ti2—Fe1 ⁱⁱ	95.19 (3)
Ti2 ^{vii} —Ti1—Ti2 ⁱⁱⁱ	62.010 (9)	Fe1 ^{xxi} —Ti2—Fe1 ⁱⁱ	95.19 (3)
Fe1 ⁱ —Ti1—Ti2 ^{vi}	121.60 (3)	C1 ^{xvi} —Ti2—Fe1	81.69 (3)
Fe1 ⁱⁱ —Ti1—Ti2 ^{vi}	58.40 (3)	C1 ^{xvii} —Ti2—Fe1	137.91 (5)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ^{vi}	115.261 (17)	Fe1 ^{xx} —Ti2—Fe1	95.19 (3)
Fe1 ^{iv} —Ti1—Ti2 ^{vi}	115.261 (17)	Fe1 ^{xxi} —Ti2—Fe1	95.19 (3)
Fe1 ^v —Ti1—Ti2 ^{vi}	64.739 (17)	Fe1 ⁱⁱ —Ti2—Fe1	56.23 (6)
Fe1 ^{vi} —Ti1—Ti2 ^{vi}	64.739 (17)	C1 ^{xvi} —Ti2—Ti1 ^{iv}	104.226 (19)
Ti2 ^{vii} —Ti1—Ti2 ^{vi}	117.990 (9)	C1 ^{xvii} —Ti2—Ti1 ^{iv}	104.226 (19)
Ti2 ⁱⁱⁱ —Ti1—Ti2 ^{vi}	180.0	Fe1 ^{xx} —Ti2—Ti1 ^{iv}	52.40 (2)
Fe1 ⁱ —Ti1—Ti2 ^{viii}	115.261 (17)	Fe1 ^{xxi} —Ti2—Ti1 ^{iv}	139.38 (5)
Fe1 ⁱⁱ —Ti1—Ti2 ^{viii}	64.739 (17)	Fe1 ⁱⁱ —Ti2—Ti1 ^{iv}	50.21 (2)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ^{viii}	64.739 (17)	Fe1—Ti2—Ti1 ^{iv}	50.21 (2)
Fe1 ^{iv} —Ti1—Ti2 ^{viii}	121.60 (3)	C1 ^{xvi} —Ti2—Ti1 ⁱⁱⁱ	104.226 (19)
Fe1 ^v —Ti1—Ti2 ^{viii}	58.40 (3)	C1 ^{xvii} —Ti2—Ti1 ⁱⁱⁱ	104.226 (19)
Fe1 ^{vi} —Ti1—Ti2 ^{viii}	115.261 (17)	Fe1 ^{xx} —Ti2—Ti1 ⁱⁱⁱ	139.38 (5)
Ti2 ^{vii} —Ti1—Ti2 ^{viii}	62.010 (9)	Fe1 ^{xxi} —Ti2—Ti1 ⁱⁱⁱ	52.40 (2)
Ti2 ⁱⁱⁱ —Ti1—Ti2 ^{viii}	117.990 (9)	Fe1 ⁱⁱ —Ti2—Ti1 ⁱⁱⁱ	50.21 (2)
Ti2 ^{vi} —Ti1—Ti2 ^{viii}	62.010 (9)	Fe1—Ti2—Ti1 ⁱⁱⁱ	50.21 (2)
Fe1 ⁱ —Ti1—Ti2 ^{ix}	64.739 (17)	Ti1 ^{iv} —Ti2—Ti1 ⁱⁱⁱ	86.98 (4)
Fe1 ⁱⁱ —Ti1—Ti2 ^{ix}	115.261 (17)	C1 ^{xvi} —Ti2—Ti2 ^{xviii}	149.47 (3)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ^{ix}	115.261 (17)	C1 ^{xvii} —Ti2—Ti2 ^{xviii}	45.23 (2)
Fe1 ^{iv} —Ti1—Ti2 ^{ix}	58.40 (3)	Fe1 ^{xx} —Ti2—Ti2 ^{xviii}	121.68 (3)
Fe1 ^v —Ti1—Ti2 ^{ix}	121.60 (3)	Fe1 ^{xxi} —Ti2—Ti2 ^{xviii}	61.47 (2)
Fe1 ^{vi} —Ti1—Ti2 ^{ix}	64.739 (17)	Fe1 ⁱⁱ —Ti2—Ti2 ^{xviii}	53.38 (3)
Ti2 ^{vii} —Ti1—Ti2 ^{ix}	117.990 (9)	Fe1—Ti2—Ti2 ^{xviii}	100.81 (3)
Ti2 ⁱⁱⁱ —Ti1—Ti2 ^{ix}	62.010 (9)	Ti1 ^{iv} —Ti2—Ti2 ^{xviii}	100.29 (4)
Ti2 ^{vi} —Ti1—Ti2 ^{ix}	117.990 (9)	Ti1 ⁱⁱⁱ —Ti2—Ti2 ^{xviii}	58.995 (4)
Ti2 ^{viii} —Ti1—Ti2 ^{ix}	180.0	C1 ^{xvi} —Ti2—Ti2 ^{xix}	149.47 (3)
Fe1 ⁱ —Ti1—Ti2 ^x	115.261 (17)	C1 ^{xvii} —Ti2—Ti2 ^{xix}	45.23 (2)
Fe1 ⁱⁱ —Ti1—Ti2 ^x	64.739 (17)	Fe1 ^{xx} —Ti2—Ti2 ^{xix}	61.47 (2)
Fe1 ⁱⁱⁱ —Ti1—Ti2 ^x	121.60 (3)	Fe1 ^{xxi} —Ti2—Ti2 ^{xix}	121.68 (3)
Fe1 ^{iv} —Ti1—Ti2 ^x	64.739 (17)	Fe1 ⁱⁱ —Ti2—Ti2 ^{xix}	53.38 (3)
Fe1 ^v —Ti1—Ti2 ^x	115.261 (17)	Fe1—Ti2—Ti2 ^{xix}	100.81 (3)
Fe1 ^{vi} —Ti1—Ti2 ^x	58.40 (3)	Ti1 ^{iv} —Ti2—Ti2 ^{xix}	58.995 (4)
Ti2 ^{vii} —Ti1—Ti2 ^x	180.0	Ti1 ⁱⁱⁱ —Ti2—Ti2 ^{xix}	100.29 (4)
Ti2 ⁱⁱⁱ —Ti1—Ti2 ^x	117.990 (9)	Ti2 ^{xviii} —Ti2—Ti2 ^{xix}	60.54 (6)
Ti2 ^{vi} —Ti1—Ti2 ^x	62.010 (9)	C1 ^{xvi} —Ti2—Ti2 ^{xi}	45.23 (2)
Ti2 ^{viii} —Ti1—Ti2 ^x	117.990 (9)	C1 ^{xvii} —Ti2—Ti2 ^{xi}	149.47 (3)
Ti2 ^{ix} —Ti1—Ti2 ^x	62.010 (9)	Fe1 ^{xx} —Ti2—Ti2 ^{xi}	121.68 (3)
Ti1 ⁱⁱ —Fe1—Ti1 ⁱⁱⁱ	108.58 (3)	Fe1 ^{xxi} —Ti2—Ti2 ^{xi}	61.47 (2)

Ti1 ⁱⁱ —Fe1—Ti1 ^{iv}	108.58 (3)	Fe1 ⁱⁱ —Ti2—Ti2 ^{xi}	100.81 (3)
Ti1 ⁱⁱⁱ —Fe1—Ti1 ^{iv}	108.58 (3)	Fe1—Ti2—Ti2 ^{xi}	53.38 (3)
Ti1 ⁱⁱ —Fe1—Ti2 ^{xi}	124.77 (2)	Ti1 ^{iv} —Ti2—Ti2 ^{xi}	100.29 (4)
Ti1 ⁱⁱⁱ —Fe1—Ti2 ^{xi}	69.20 (3)	Ti1 ⁱⁱⁱ —Ti2—Ti2 ^{xi}	58.995 (4)
Ti1 ^{iv} —Fe1—Ti2 ^{xi}	124.77 (2)	Ti2 ^{xviii} —Ti2—Ti2 ^{xi}	112.60 (3)
Ti1 ⁱⁱ —Fe1—Ti2 ^{xii}	124.77 (2)	Ti2 ^{xix} —Ti2—Ti2 ^{xi}	153.19 (5)
Ti1 ⁱⁱⁱ —Fe1—Ti2 ^{xii}	124.77 (2)	C1 ^{xvi} —Ti2—Ti2 ^{xii}	45.23 (2)
Ti1 ^{iv} —Fe1—Ti2 ^{xii}	69.20 (3)	C1 ^{xvii} —Ti2—Ti2 ^{xii}	149.47 (3)
Ti2 ^{xi} —Fe1—Ti2 ^{xii}	69.49 (5)	Fe1 ^{xx} —Ti2—Ti2 ^{xii}	61.47 (2)
Ti1 ⁱⁱ —Fe1—Ti2 ^{xiii}	69.20 (3)	Fe1 ^{xxi} —Ti2—Ti2 ^{xii}	121.68 (3)
Ti1 ⁱⁱⁱ —Fe1—Ti2 ^{xiii}	124.77 (2)	Fe1 ⁱⁱ —Ti2—Ti2 ^{xii}	100.81 (3)
Ti1 ^{iv} —Fe1—Ti2 ^{xiii}	124.77 (2)	Fe1—Ti2—Ti2 ^{xii}	53.38 (3)
Ti2 ^{xi} —Fe1—Ti2 ^{xiii}	69.49 (5)	Ti1 ^{iv} —Ti2—Ti2 ^{xii}	58.995 (4)
Ti2 ^{xii} —Fe1—Ti2 ^{xiii}	69.49 (5)	Ti1 ⁱⁱⁱ —Ti2—Ti2 ^{xii}	100.29 (4)
Ti1 ⁱⁱ —Fe1—Fe1 ⁱⁱⁱ	56.32 (2)	Ti2 ^{xviii} —Ti2—Ti2 ^{xii}	153.19 (5)
Ti1 ⁱⁱⁱ —Fe1—Fe1 ⁱⁱⁱ	104.92 (3)	Ti2 ^{xix} —Ti2—Ti2 ^{xii}	112.60 (3)
Ti1 ^{iv} —Fe1—Fe1 ⁱⁱⁱ	56.32 (2)	Ti2 ^{xi} —Ti2—Ti2 ^{xii}	60.54 (6)
Ti2 ^{xi} —Fe1—Fe1 ⁱⁱⁱ	174.11 (3)	Ti2 ^{xxii} —C1—Ti2 ^{xxiii}	180.0
Ti2 ^{xii} —Fe1—Fe1 ⁱⁱⁱ	115.14 (3)	Ti2 ^{xxii} —C1—Ti2 ^{xxiv}	89.54 (5)
Ti2 ^{xiii} —Fe1—Fe1 ⁱⁱⁱ	115.14 (3)	Ti2 ^{xxiii} —C1—Ti2 ^{xxiv}	90.46 (5)
Ti1 ⁱⁱ —Fe1—Fe1 ⁱⁱ	104.92 (3)	Ti2 ^{xxii} —C1—Ti2 ^{xxv}	90.46 (5)
Ti1 ⁱⁱⁱ —Fe1—Fe1 ⁱⁱ	56.32 (2)	Ti2 ^{xxiii} —C1—Ti2 ^{xxv}	89.54 (5)
Ti1 ^{iv} —Fe1—Fe1 ⁱⁱ	56.32 (2)	Ti2 ^{xxiv} —C1—Ti2 ^{xxv}	180.0
Ti2 ^{xi} —Fe1—Fe1 ⁱⁱ	115.14 (3)	Ti2 ^{xxii} —C1—Ti2 ^{xxvi}	90.46 (5)
Ti2 ^{xii} —Fe1—Fe1 ⁱⁱ	115.14 (3)	Ti2 ^{xxiii} —C1—Ti2 ^{xxvi}	89.54 (5)
Ti2 ^{xiii} —Fe1—Fe1 ⁱⁱ	174.11 (3)	Ti2 ^{xxiv} —C1—Ti2 ^{xxvi}	89.54 (5)
Fe1 ⁱⁱⁱ —Fe1—Fe1 ⁱⁱ	60.0	Ti2 ^{xxv} —C1—Ti2 ^{xxvi}	90.46 (5)
Ti1 ⁱⁱ —Fe1—Fe1 ^{iv}	56.32 (2)	Ti2 ^{xxii} —C1—Ti2 ^{xxvii}	89.54 (5)
Ti1 ⁱⁱⁱ —Fe1—Fe1 ^{iv}	56.32 (2)	Ti2 ^{xxiii} —C1—Ti2 ^{xxvii}	90.46 (5)
Ti1 ^{iv} —Fe1—Fe1 ^{iv}	104.92 (3)	Ti2 ^{xxiv} —C1—Ti2 ^{xxvii}	90.46 (5)
Ti2 ^{xi} —Fe1—Fe1 ^{iv}	115.14 (3)	Ti2 ^{xxv} —C1—Ti2 ^{xxvii}	89.54 (5)
Ti2 ^{xii} —Fe1—Fe1 ^{iv}	174.11 (3)	Ti2 ^{xxvi} —C1—Ti2 ^{xxvii}	180.0
Ti2 ^{xiii} —Fe1—Fe1 ^{iv}	115.14 (3)	Ti2 ^{xxii} —O1—Ti2 ^{xxiii}	180.0
Fe1 ⁱⁱⁱ —Fe1—Fe1 ^{iv}	60.0	Ti2 ^{xxii} —O1—Ti2 ^{xxiv}	89.54 (5)
Fe1 ⁱⁱ —Fe1—Fe1 ^{iv}	60.0	Ti2 ^{xxiii} —O1—Ti2 ^{xxiv}	90.46 (5)
Ti1 ⁱⁱ —Fe1—Ti2	166.80 (5)	Ti2 ^{xxii} —O1—Ti2 ^{xxv}	90.46 (5)
Ti1 ⁱⁱⁱ —Fe1—Ti2	65.047 (6)	Ti2 ^{xxiii} —O1—Ti2 ^{xxv}	89.54 (5)
Ti1 ^{iv} —Fe1—Ti2	65.047 (6)	Ti2 ^{xxiv} —O1—Ti2 ^{xxv}	180.0
Ti2 ^{xi} —Fe1—Ti2	65.152 (14)	Ti2 ^{xxii} —O1—Ti2 ^{xxvi}	90.46 (5)
Ti2 ^{xii} —Fe1—Ti2	65.152 (14)	Ti2 ^{xxiii} —O1—Ti2 ^{xxvi}	89.54 (5)
Ti2 ^{xiii} —Fe1—Ti2	124.00 (6)	Ti2 ^{xxiv} —O1—Ti2 ^{xxvi}	89.54 (5)
Fe1 ⁱⁱⁱ —Fe1—Ti2	112.83 (2)	Ti2 ^{xxv} —O1—Ti2 ^{xxvi}	90.46 (5)
Fe1 ⁱⁱ —Fe1—Ti2	61.89 (3)	Ti2 ^{xxii} —O1—Ti2 ^{xxvii}	89.54 (5)
Fe1 ^{iv} —Fe1—Ti2	112.83 (2)	Ti2 ^{xxiii} —O1—Ti2 ^{xxvii}	90.46 (5)
Ti1 ⁱⁱ —Fe1—Ti2 ^{xiv}	65.047 (6)	Ti2 ^{xxiv} —O1—Ti2 ^{xxvii}	90.46 (5)

Ti1 ⁱⁱⁱ —Fe1—Ti2 ^{xiv}	65.047 (6)	Ti2 ^{xxv} —O1—Ti2 ^{xxvii}	89.54 (5)
Ti1 ^{iv} —Fe1—Ti2 ^{xiv}	166.80 (5)	Ti2 ^{xxvi} —O1—Ti2 ^{xxvii}	180.0

Symmetry codes: (i) $-x, y-1/4, z-1/4$; (ii) $x, -y+1/4, -z+1/4$; (iii) $-x+1/4, -y+1/4, z$; (iv) $-x+1/4, y, -z+1/4$; (v) $x-1/4, -y, z-1/4$; (vi) $x-1/4, y-1/4, -z$; (vii) $y-1/4, -z, x-1/4$; (viii) $z, -x+1/4, -y+1/4$; (ix) $-z, x-1/4, y-1/4$; (x) $-y+1/4, z, -x+1/4$; (xi) $y+1/4, -z+1/2, x-1/4$; (xii) $-z+1/2, x-1/4, y+1/4$; (xiii) $x-1/4, y+1/4, -z+1/2$; (xiv) z, x, y ; (xv) y, z, x ; (xvi) $x, -y+3/4, -z+3/4$; (xvii) $x, y-1/2, z-1/2$; (xviii) $-z+1/2, -x+1/2, -y$; (xix) $-y+1/2, -z, -x+1/2$; (xx) $x+1/4, y-1/4, -z+1/2$; (xxi) $x+1/4, -y+1/2, z-1/4$; (xxii) $-z+1/2, -x+1, -y+1/2$; (xxiii) $z+1/2, x, y+1/2$; (xxiv) $x, y+1/2, z+1/2$; (xxv) $-x+1, -y+1/2, -z+1/2$; (xxvi) $-y+1/2, -z+1/2, -x+1$; (xxvii) $y+1/2, z+1/2, x$.