

Acta Crystallographica Section E Structure Reports Online

ISSN 1600-5368

## Calcium platinum aluminium, CaPtAl

#### Patrice Kenfack Tsobnang,<sup>a</sup> Daniel Fotio,<sup>b</sup> Siméon Ponou<sup>c</sup>\* and Charles Fon Abi<sup>c</sup>

<sup>a</sup>The University of Yaounde 1, Department of Inorganic Chemistry, Faculty of Sciences, PO Box 812 Yaounde, Cameroon, <sup>b</sup>The University of Maroua, Department of Chemistry, Ecole Normale Supérieure de Maroua, Cameroon, and <sup>c</sup>The University of Yaounde 1, Department of Chemistry, Ecole Normale Supérieure de Yaoundé, PO Box 47 Yaounde, Cameroon

Correspondence e-mail: sponou@gmail.com

Received 5 September 2011; accepted 9 September 2011

Key indicators: single-crystal X-ray study; T = 293 K; mean  $\sigma$ (Pt–Al) = 0.003 Å; R factor = 0.028; wR factor = 0.064; data-to-parameter ratio = 25.5.

A preliminary X-ray study of CaPtAl has been reported previously by Hulliger [*J. Alloys Compd* (1993), **196**, 225–228] based on X-ray powder diffraction data without structure refinement. With the present single-crystal X-ray study, we confirm the assignment of the TiNiSi type for CaPtAl, in a fully ordered inverse structure. All three atoms of the asymmetric unit have *.m.* site symmetry. The structure features a  $^{3}_{\infty}$ [AlPt] open framework with a fourfold coordination of Pt by Al atoms and *vice versa*. The Ca atoms are located in the large channels of the structure.

#### **Related literature**

For a previous X-ray powder diffraction study of CaPtAl, see: Hulliger (1993). For related compounds, see: Dascoulidou-Gritner & Schuster (1994); Merlo *et al.* (1996). For structural systematics and properties of the TiNiSi structure type, see: Kussmann *et al.* (1998); Hoffmann & Pöttgen (2001); Nuspl *et al.* (1996); Evers *et al.* (1992). For related compounds of the TiNiSi structure type, see: Ponou & Lidin (2008); Ponou (2010); Banenzoué *et al.* (2009). For atomic radii, see: Pauling (1960).

#### **Experimental**

Crystal data CaPtAl  $M_r = 262.15$ 

Orthorhombic, *Pnma* a = 7.1581 (14) Å b = 4.2853 (15) Åc = 7.7536 (9) Å $V = 237.84 (10) \text{ Å}^3$ Z = 4

#### Data collection

Oxford Diffraction Xcalibur diffractometer with Sapphire3 CCD Absorption correction: multi-scan (*CrysAlis RED*; Oxford

Refinement

 $R[F^2 > 2\sigma(F^2)] = 0.028$  $wR(F^2) = 0.064$ S = 1.03511 reflections Mo  $K\alpha$  radiation  $\mu = 61.08 \text{ mm}^{-1}$  T = 293 K $0.15 \times 0.08 \times 0.05 \text{ mm}$ 

Diffraction, 2007)  $T_{\min} = 0.004$ ,  $T_{\max} = 0.047$ 4413 measured reflections 511 independent reflections 435 reflections with  $I > 2\sigma(I)$  $R_{\text{int}} = 0.091$ 

20 parameters  $\Delta \rho_{\text{max}} = 2.61 \text{ e } \text{ Å}^{-3}$  $\Delta \rho_{\text{min}} = -2.95 \text{ e } \text{ Å}^{-3}$ 

Data collection: *CrysAlis CCD* (Oxford Diffraction, 2007); cell refinement: *CrysAlis RED* (Oxford Diffraction, 2007); data reduction: *CrysAlis CCD*; program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: *DIAMOND* (Brandenburg, 1999); software used to prepare material for publication: *WinGX* (Farrugia, 1999).

This work was supported financially by the University of Lund (Sweden). SP thanks Professor Sven Lidin (Division of Material Chemistry, Lund University) for continuing support.

Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2532).

#### References

Banenzoué, C., Ponou, S. & Lambi, J. N. (2009). Acta Cryst. E65, i90.

- Brandenburg, K. (1999). *DIAMOND*. Crystal Impact GbR, Bonn, Germany. Dascoulidou-Gritner, K. & Schuster, H. U. (1994). Z. Anorg. Allg. Chem. 620, 1151–1156.
- Evers, J., Oehlinger, G., Polborn, K. & Sendlinger, B. (1992). J. Alloys Compd, 182, L23–L29.
- Farrugia, L. J. (1999). J. Appl. Cryst. 32, 837-838.
- Hoffmann, R.-D. & Pöttgen, R. (2001). Z. Kristallogr. 216, 127-145.
- Hulliger, F. (1993). J. Alloys Compd, 196, 225-228.
- Kussmann, D., Hoffmann, R.-D. & Pöttgen, R. (1998). Z. Anorg. Allg. Chem. 624, 1727–1735.

Merlo, F., Pani, M. & Fornasini, M. L. (1996). J. Alloys Compd, 232, 289–295. Nuspl, G., Polborn, K., Evers, J., Landrum, G. A. & Hoffmann, R. (1996).

- Nuspi, G., Poloorn, K., Evers, J., Landrum, G. A. & Horimann, K. (1996). Inorg. Chem. **35**, 6922–6932.
- Oxford Diffraction (2007). CrysAlis CCD and CrysAlis RED. Oxford Diffraction Ltd, Abingdon, England
- Pauling, L. (1960). The Nature of the Chemical Bond, 3rd ed. Ithaca: Cornell University Press.
- Ponou, S. (2010). Eur. J. Inorg. Chem., pp. 4139-4147.
- Ponou, S. & Lidin, S. (2008). Z. Kristallogr. New Cryst. Struct. 223, 329–330. Sheldrick, G. M. (2008). Acta Cryst. A64, 112–122.

# supporting information

Acta Cryst. (2011). E67, i55 [https://doi.org/10.1107/S1600536811036749]

### Calcium platinum aluminium, CaPtAl

### Patrice Kenfack Tsobnang, Daniel Fotio, Siméon Ponou and Charles Fon Abi

#### S1. Comment

A recent re-investigation of the ternary CaAgGe phase from single-crystal X-ray data has revealed a superstructure with a tripling of the *a*-axis of the of the TiNiSi-type basic cell (*i*<sub>3</sub>, minimal isomorphic subgroup of index 3) and a complete ordering of the atomic sites (Ponou & Lidin, 2008; Ponou, 2010). This phase has previously been reported by Merlo *et al.* (1996) in the KHg<sub>2</sub> structure type with Ag and Ge atoms randomly distributed on the Hg site. Hence, precice single crystals X-ray diffraction measurements is of importance in this family of compounds (Kussmann *et al.*, 1998; Banenzoué *et al.*, 2009) to access both the superstructure and a possible structure inversion with respect to the Ni and Si positions in the three-dimensional framework of fourfold interconnected atoms. Such a structure inversion is observed, for example, in CaGaPt, with Ga atoms at the orginal Ni position and Pt at the original Si position with respect to the prototype TiNiSi structure type (Dascoulidou-Gritner & Schuster, 1994). In 'normal' TiNiSi phases like CaPtGe, Ge occupies the Si position and Pt the Ni position (Evers *et al.*, 1992; Nuspl *et al.*, 1996). Here we report the structure refinement of CaPtAl from single-crystal X-ray diffraction data.

The structure of the title compound was first investigated by Hulliger (1993) from X-ray powder data and assigned to the orthorhombic TiNiSi structure type, however, without a detailed structural refinement. In the present study, the CaPtAl structure was successfully refined as an inverse TiNiSi type. Hence, Ca, Al and Pt atoms are found at Ti, Ni and Si atomic positions, respectively. The origin of the inversion in this structure type is generally ascribed to the relative electronegativity of the framework constituent elements, here Pt and Al. The more electronegative atom (here Pt) is found at the Si position in a strongly distorted tetrahedral (rather pyramidal) coordination whereas the coordination of the less electronegative atom at the Ni position is only slightly distorted from its idealized tetrahedral values (Hoffmann & Pöttgen, 2001). The Pt—Al interactomic distances range from 2.574 (3) Å to 2.675 (3) Å which is comparable with the sum of atomic radii of these elements, *i.e.* 2.62 Å (Al: 1.25 and Pt: 1.37 Å; Pauling, 1960), indicating a covalent character of the bonding. The first coordination sphere of the Ca atoms consists of two counter-tilted Pt<sub>3</sub>Al<sub>3</sub> hexagons with Ca—Al and Ca—Pt distances ranging from 3.143 (3) Å to 3.489 (3) Å and from 2.978 (2) Å to 3.1179 (16) Å, respectively, also in agreement with the sum of atomic radii (Ca: 1.97 Å).

#### **S2. Experimental**

Single crystals of the title compound, suitable for X-ray diffraction studies, were obtained from a mixture of the elements (Ca ingots (99.5%), Al chunk (99.9%) and Pt pieces (99%), all from ABCR) with a molar ratio Ca:Pt:Al = 2:2:1, by heating in a Nb ampoule at 1253 K for one hour and then slowly cooling at a rate of 6 K/h to room temperature. The product is air-stable, silver-grey with metallic lustre.

#### **S3. Refinement**

The refined unit cell parameters are quite close (slighly lower) to those obtained from a Guinier camera by Hulliger (1993) with a = 7.1722 (6), b = 4.2885 (4) and c = 7.7760 (7) Å. The refinement in the inverse TiNiSi structure model was straightforward. Full/mixed occupancies were checked for all atomic sites by freeing the site occupation factor for each given individual atom, while keeping that of the other atoms fixed. The residual map shows highest peak/deepest hole of 2.61/-2.95 e Å<sup>-2</sup> at 0.75/0.60 Å from Pt1, respectively.



#### Figure 1

A perspective view of the CaPtAl structure with displacement ellipsoids drawn at the 99% probability level. Ca, Al, and Pt atoms are drawn as white-crossed, grey and black spheres, respectively.

#### Calcium aluminium platinum

Crystal d	lata
-----------	------

CaPtAl  $M_r = 262.15$ Orthorhombic, *Pnma* Hall symbol: -P 2ac 2n a = 7.1581 (14) Å b = 4.2853 (15) Å c = 7.7536 (9) Å V = 237.84 (10) Å<sup>3</sup> Z = 4

#### Data collection

Oxford Diffraction Xcalibur diffractometer with Sapphire3 CCD Graphite monochromator F(000) = 444  $D_x = 7.321 \text{ Mg m}^{-3}$ Mo K\alpha radiation, \lambda = 0.71073 \mathcal{A} Cell parameters from 2105 reflections  $\theta = 2.6-33.7^{\circ}$   $\mu = 61.08 \text{ mm}^{-1}$  T = 293 KIrregular block, metallic grey  $0.15 \times 0.08 \times 0.05 \text{ mm}$ 

Detector resolution: 16.1829 pixels mm<sup>-1</sup>  $\omega$  and  $\varphi$  scans

Absorption correction: multi-scan	$R_{\rm int} = 0.091$
(CrysAlis RED; Oxford Diffraction, 2007)	$\theta_{\rm max} = 33.8^\circ, \ \theta_{\rm min} = 3.9^\circ$
$T_{\min} = 0.004, \ T_{\max} = 0.047$	$h = -10 \rightarrow 11$
4413 measured reflections	$k = -6 \rightarrow 6$
511 independent reflections	$l = -11 \rightarrow 12$
435 reflections with $I > 2\sigma(I)$	
Refinement	

Secondary atom site location: difference Fourier
map
$w = 1/[\sigma^2(F_o^2) + (0.0326P)^2]$
where $P = (F_o^2 + 2F_c^2)/3$
$(\Delta/\sigma)_{\rm max} < 0.001$
$\Delta \rho_{\rm max} = 2.61 \text{ e } \text{\AA}^{-3}$
$\Delta \rho_{\rm min} = -2.95 \text{ e } \text{\AA}^{-3}$
Extinction correction: SHELXL97 (Sheldrick,
2008)
Extinction coefficient: 0.0027 (5)

#### Special details

**Geometry**. All e.s.d.'s (except the e.s.d. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell e.s.d.'s are taken into account individually in the estimation of e.s.d.'s in distances, angles and torsion angles; correlations between e.s.d.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell e.s.d.'s is used for estimating e.s.d.'s involving l.s. planes.

**Refinement**. Refinement of  $F^2$  against ALL reflections. The weighted *R*-factor *wR* and goodness of fit *S* are based on  $F^2$ , conventional *R*-factors *R* are based on *F*, with *F* set to zero for negative  $F^2$ . The threshold expression of  $F^2 > \sigma(F^2)$  is used only for calculating *R*-factors(gt) *etc.* and is not relevant to the choice of reflections for refinement. *R*-factors based on  $F^2$  are statistically about twice as large as those based on *F*, and *R*- factors based on ALL data will be even larger.

	x	У	Ζ	$U_{ m iso}$ */ $U_{ m eq}$
Pt1	0.78928 (5)	0.25	0.38355 (4)	0.01029 (15)
Al1	0.1446 (5)	0.25	0.4346 (4)	0.0103 (6)
Cal	0.4796 (3)	0.75	0.3234 (2)	0.0106 (4)

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters  $(Å^2)$ 

Atomic displacement parameters  $(Å^2)$ 

	<i>U</i> <sup>11</sup>	<i>U</i> <sup>22</sup>	<i>U</i> <sup>33</sup>	$U^{12}$	U <sup>13</sup>	<i>U</i> <sup>23</sup>
Pt1	0.00775 (19)	0.0118 (2)	0.0113 (2)	0	0.00104 (12)	0
A11	0.0087 (13)	0.0127 (15)	0.0094 (11)	0	0.0001 (11)	0
Ca1	0.0088 (8)	0.0129 (10)	0.0102 (8)	0	-0.0011 (6)	0

#### Geometric parameters (Å, °)

Pt1—Al1 <sup>i</sup>	2.574 (3)	Al1—Ca1 <sup>xi</sup>	3.160 (3)
Pt1—Al1 <sup>ii</sup>	2.6084 (17)	Al1—Ca1 <sup>ix</sup>	3.160 (3)
Pt1—Al1 <sup>iii</sup>	2.6084 (17)	Al1—Ca1 <sup>ii</sup>	3.280 (4)
Pt1—Al1 <sup>iv</sup>	2.675 (3)	Al1—Ca1	3.329 (3)
Pt1—Ca1 <sup>ii</sup>	2.978 (2)	Al1—Ca1 <sup>vi</sup>	3.329 (3)
Pt1—Ca1 <sup>v</sup>	3.0037 (14)	Ca1—Pt1 <sup>ii</sup>	2.978 (2)

# supporting information

Pt1—Ca1 <sup>iv</sup>	3 0037 (14)	Ca1—Pt1 <sup>ix</sup>	3 0037 (14)
Pt1—Ca1 <sup>vi</sup>	3 1179 (16)	$Ca1 - Pt1^{xii}$	3 0037 (14)
Pt1—Ca1	3 1179 (16)	$Ca1 - Pt1^{xiii}$	3.1179 (16)
$Pt1 - Ca1^{vii}$	3 791 (2)	$Ca1 = A11x^{iv}$	3.1175(10)
$A11  \mathbf{Pt} 1^{\text{viii}}$	2.771(2)		3.145(3)
A11 - Dt1	2.574(5)		3.160(3)
A11 - D t 1	2.0084(17)	Ca1 = A11	3.100(3)
A11 - D t 1 ix	2.0084(17)	Ca1 = A11	3.280(4)
A11 - Co1x	2.075(3) 2.142(2)	Cal—All	3.329(3)
All—Cal"	5.145 (5)	Cal—Cal····	5.469 (5)
All <sup>i</sup> —Pt1—All <sup>ii</sup>	74.79 (9)	Ca1 <sup>xi</sup> —Al1—Ca1	122.80 (9)
All <sup>i</sup> —Pt1—All <sup>iii</sup>	74.79 (8)	Ca1 <sup>ix</sup> —Al1—Ca1	70.67 (4)
All <sup>ii</sup> —Pt1—All <sup>iii</sup>	110.46 (10)	Ca1 <sup>ii</sup> —Al1—Ca1	63.73 (7)
$A11^{i}$ $Pt1$ $A11^{iv}$	121.62 (9)	Pt1 <sup>viii</sup> —A11—Ca1 <sup>vi</sup>	132.22 (7)
$A11^{ii}$ $Pt1$ $A11^{iv}$	124.66 (5)	Pt1 <sup>ii</sup> —A11—Ca1 <sup>vi</sup>	122.54 (12)
$A11^{iii}$ $Pt1$ $A11^{iv}$	124.66 (5)	$Pt1^{iii}$ Al1 — Ca1 <sup>vi</sup>	58.71 (5)
$A11^{i}$ $Pt1$ $Ca1^{ii}$	121.43 (7)	$Pt1^{ix}$ Al1 — Ca1 <sup>vi</sup>	58.83 (6)
$A11^{ii}$ $Pt1$ $Ca1^{ii}$	72.83 (7)	$Ca1^{x}$ All $Ca1^{vi}$	116 88 (8)
$A11^{iii} - Pt1 - Ca1^{ii}$	72.83 (7)	$Ca1^{xi}$ All $Ca1^{vi}$	70 67 (4)
$A11^{iv} Pt1 Ca1^{ii}$	116.96 (8)	$Ca1^{ix}$ $A11$ $Ca1^{vi}$	122 80 (9)
$A11^{i} Pt1 Ca1^{v}$	68 52 (5)	$Ca1^{ii}$ $A11$ $Ca1^{vi}$	6373(7)
$\Delta 11^{ii} - Pt1 - Ca1^{v}$	142 42 (8)	$Ca1 = A11 = Ca1^{vi}$	80.12 (9)
$\Delta 11^{iii} - Pt1 - Ca1^{v}$	67.70(7)	$Pt_{1i} = Ca_{1} = Pt_{1ix}$	96 57 (5)
$A11^{iv} Pt1 Ca1^{v}$	71.52 (6)	$Pt1^{ii}$ Col $Pt1^{xii}$	96.57 (5)
AII - III - CaI	124.40(2)	$\begin{array}{ccc} \mathbf{P} \mathbf{T} \mathbf{T} \mathbf{T} \mathbf{T} \mathbf{T} \mathbf{T} \mathbf{T} T$	90.37 (3)
$A_{11i} = D_{t1} = C_{a1iv}$	134.49(3)	$\mathbf{P}_{\mathbf{I}} = \mathbf{P}_{\mathbf{I}} = \mathbf{P}_{\mathbf{I}} = \mathbf{P}_{\mathbf{I}} = \mathbf{P}_{\mathbf{I}} = \mathbf{P}_{\mathbf{I}}$	91.01(0)
A11 - Ft1 - Ca1	(00.32(3))	$\mathbf{r}_{t1} = \mathbf{C}_{a1} = \mathbf{r}_{t1}$ $\mathbf{p}_{t1} = \mathbf{r}_{a1} = \mathbf{p}_{t1}$	110.21(3)
All $\frac{1}{10}$ PtI Cali	0/./0(/)	$Pt1^{xy} = Ca1 = Pt1^{xy}$	133.20(7)
Alliv Dtl Cali	142.42(6)	$\mathbf{P}(\mathbf{I}^{\text{min}} - \mathbf{C}\mathbf{a}^{\text{min}} - \mathbf{P}(\mathbf{I}^{\text{min}})$	84.90 (3)
AII <sup></sup> —PII—CaI <sup></sup>	/1.52 (0)	PII <sup>T</sup> Cal PII	110.21(5)
$Cal^{\mu}$ —Ptl—Cal <sup><math>\mu</math></sup>	134.49 (3)	$PtI^{\pm}$ —CaI—PtI	84.96 (3)
Cal <sup>*</sup> —Ptl—Cal <sup>*</sup>	91.01 (6)	Pt1 <sup>xii</sup> —Ca1—Pt1	153.20(7)
	136.51 (3)	$PtI^{xm}$ —CaI—PtI	86.82 (5)
All <sup>m</sup> —Ptl—Cal <sup>m</sup>	140.75 (8)	$PtI^{n}$ —Cal—All <sup>xiv</sup>	123.30 (9)
$AII^{m}$ — $PtI$ — $CaI^{vi}$	69.23 (7)	Pt1 <sup>ix</sup> —Ca1—A11 <sup>xiv</sup>	50.15 (4)
$AII^{W}$ $PtI$ $CaI^{W}$	65.60 (6)	$PtI^{xii}$ —Cal—All <sup>xiv</sup>	50.15 (4)
$Ca1^n$ —Pt1—Ca1 <sup>v1</sup>	69.79 (5)	Pt1 <sup>xm</sup> —Ca1—Al1 <sup>xn</sup>	110.15 (6)
$Ca1^v$ —Pt1—Ca1 <sup>v1</sup>	75.66 (3)	Pt1—Ca1—Al1 <sup>xiv</sup>	110.15 (6)
Cal <sup>1v</sup> —Pt1—Cal <sup>v1</sup>	137.12 (3)	Pt1 <sup>n</sup> —Ca1—Al1 <sup>w</sup>	136.43 (5)
Alli—Pt1—Cal	136.51 (3)	Pt1 <sup>ix</sup> —Ca1—Al1 <sup>iv</sup>	49.29 (6)
All <sup>ii</sup> —Pt1—Ca1	69.23 (7)	Pt1 <sup>xii</sup> —Ca1—Al1 <sup>iv</sup>	108.36 (7)
All <sup>iii</sup> —Pt1—Ca1	140.74 (8)	Pt1 <sup>xiii</sup> —Ca1—Al1 <sup>iv</sup>	107.15 (8)
All <sup>iv</sup> —Pt1—Ca1	65.60 (6)	Pt1—Ca1—Al1 <sup>iv</sup>	50.44 (6)
Cal <sup>ii</sup> —Pt1—Cal	69.79 (5)	All <sup>xiv</sup> —Cal—All <sup>iv</sup>	59.91 (8)
Ca1 <sup>v</sup> —Pt1—Ca1	137.12 (3)	Pt1 <sup>ii</sup> —Ca1—A11 <sup>xv</sup>	136.43 (5)
Cal <sup>iv</sup> —Pt1—Cal	75.66 (3)	Pt1 <sup>ix</sup> —Ca1—Al1 <sup>xv</sup>	108.36 (7)
Cal <sup>vi</sup> —Pt1—Cal	86.82 (5)	Pt1 <sup>xii</sup> —Ca1—Al1 <sup>xv</sup>	49.29 (6)
Alli—Pt1—Cal <sup>vii</sup>	55.28 (7)	Pt1 <sup>xiii</sup> —Ca1—Al1 <sup>xv</sup>	50.44 (6)
All <sup>ii</sup> —Pt1—Ca1 <sup>vii</sup>	55.55 (5)	Pt1—Ca1—Al1 <sup>xv</sup>	107.15 (8)

All <sup>iii</sup> —Pt1—Ca1 <sup>vii</sup>	55.55 (5)	All <sup>xiv</sup> —Cal—All <sup>xv</sup>	59.91 (8)
All <sup>iv</sup> —Pt1—Ca1 <sup>vii</sup>	176.90 (8)	All <sup>iv</sup> —Cal—All <sup>xv</sup>	85.38 (9)
Ca1 <sup>ii</sup> —Pt1—Ca1 <sup>vii</sup>	66.144 (17)	Pt1 <sup>ii</sup> —Ca1—Al1 <sup>ii</sup>	95.37 (7)
Ca1 <sup>v</sup> —Pt1—Ca1 <sup>vii</sup>	106.42 (4)	Pt1 <sup>ix</sup> —Ca1—Al1 <sup>ii</sup>	132.66 (3)
Ca1 <sup>iv</sup> —Pt1—Ca1 <sup>vii</sup>	106.42 (4)	Pt1 <sup>xii</sup> —Ca1—Al1 <sup>ii</sup>	132.66 (3)
Ca1 <sup>vi</sup> —Pt1—Ca1 <sup>vii</sup>	116.41 (5)	Pt1 <sup>xiii</sup> —Ca1—Al1 <sup>ii</sup>	48.04 (3)
Ca1—Pt1—Ca1 <sup>vii</sup>	116.41 (5)	Pt1—Ca1—Al1 <sup>ii</sup>	48.04 (3)
Pt1 <sup>viii</sup> —Al1—Pt1 <sup>ii</sup>	105.21 (8)	Al1 <sup>xiv</sup> —Ca1—Al1 <sup>ii</sup>	141.34 (12)
Pt1 <sup>viii</sup> —Al1—Pt1 <sup>iii</sup>	105.21 (8)	All <sup>iv</sup> —Cal—All <sup>ii</sup>	93.19 (5)
Pt1 <sup>ii</sup> —Al1—Pt1 <sup>iii</sup>	110.46 (10)	Al1 <sup>xv</sup> —Ca1—Al1 <sup>ii</sup>	93.19 (5)
Pt1 <sup>viii</sup> —Al1—Pt1 <sup>ix</sup>	103.93 (10)	Pt1 <sup>ii</sup> —Ca1—Al1	48.46 (5)
Pt1 <sup>ii</sup> —Al1—Pt1 <sup>ix</sup>	115.36 (8)	Pt1 <sup>ix</sup> —Ca1—Al1	49.65 (5)
Pt1 <sup>iii</sup> —Al1—Pt1 <sup>ix</sup>	115.36 (8)	Pt1 <sup>xii</sup> —Ca1—Al1	105.70 (8)
Pt1 <sup>viii</sup> —Al1—Ca1 <sup>x</sup>	82.41 (9)	Pt1 <sup>xiii</sup> —Ca1—Al1	156.29 (8)
Pt1 <sup>ii</sup> —Al1—Ca1 <sup>x</sup>	62.14 (6)	Pt1—Ca1—Al1	91.78 (5)
Pt1 <sup>iii</sup> —Al1—Ca1 <sup>x</sup>	62.14 (6)	All <sup>xiv</sup> —Ca1—All	92.55 (6)
Pt1 <sup>ix</sup> —Al1—Ca1 <sup>x</sup>	173.66 (14)	All <sup>iv</sup> —Cal—All	89.81 (6)
Pt1 <sup>viii</sup> —Al1—Ca1 <sup>xi</sup>	62.19 (7)	All <sup>xv</sup> —Ca1—Al1	150.39 (11)
Pt1 <sup>ii</sup> —Al1—Ca1 <sup>xi</sup>	165.23 (11)	All <sup>ii</sup> —Cal—All	116.27 (7)
Pt1 <sup>iii</sup> —Al1—Ca1 <sup>xi</sup>	81.55 (4)	Pt1 <sup>ii</sup> —Ca1—Al1 <sup>xiii</sup>	48.46 (5)
Pt1 <sup>ix</sup> —Al1—Ca1 <sup>xi</sup>	63.96 (6)	Pt1 <sup>ix</sup> —Ca1—Al1 <sup>xiii</sup>	105.70 (8)
Ca1 <sup>x</sup> —Al1—Ca1 <sup>xi</sup>	120.09 (8)	Pt1 <sup>xii</sup> —Ca1—Al1 <sup>xiii</sup>	49.65 (5)
Pt1 <sup>viii</sup> —Al1—Ca1 <sup>ix</sup>	62.19 (7)	Pt1 <sup>xiii</sup> —Ca1—Al1 <sup>xiii</sup>	91.78 (5)
Pt1 <sup>ii</sup> —Al1—Ca1 <sup>ix</sup>	81.55 (4)	Pt1—Ca1—Al1 <sup>xiii</sup>	156.29 (8)
Pt1 <sup>iii</sup> —Al1—Ca1 <sup>ix</sup>	165.23 (11)	Al1 <sup>xiv</sup> —Ca1—Al1 <sup>xiii</sup>	92.55 (6)
Pt1 <sup>ix</sup> —Al1—Ca1 <sup>ix</sup>	63.96 (6)	All <sup>iv</sup> —Ca1—All <sup>xiii</sup>	150.39 (10)
Ca1 <sup>x</sup> —Al1—Ca1 <sup>ix</sup>	120.09 (8)	Al1 <sup>xv</sup> —Ca1—Al1 <sup>xiii</sup>	89.81 (6)
Ca1 <sup>xi</sup> —Al1—Ca1 <sup>ix</sup>	85.38 (9)	Al1 <sup>ii</sup> —Ca1—Al1 <sup>xiii</sup>	116.27 (7)
Pt1 <sup>viii</sup> —Al1—Ca1 <sup>ii</sup>	153.94 (11)	Al1—Ca1—Al1 <sup>xiii</sup>	80.12 (9)
Pt1 <sup>ii</sup> —Al1—Ca1 <sup>ii</sup>	62.73 (6)	Pt1 <sup>ii</sup> —Ca1—Ca1 <sup>xvi</sup>	56.99 (5)
Pt1 <sup>iii</sup> —Al1—Ca1 <sup>ii</sup>	62.73 (6)	Pt1 <sup>ix</sup> —Ca1—Ca1 <sup>xvi</sup>	153.54 (10)
Pt1 <sup>ix</sup> —Al1—Ca1 <sup>ii</sup>	102.13 (11)	Pt1 <sup>xii</sup> —Ca1—Ca1 <sup>xvi</sup>	91.10 (3)
Ca1 <sup>x</sup> —Al1—Ca1 <sup>ii</sup>	71.53 (6)	Pt1 <sup>xiii</sup> —Ca1—Ca1 <sup>xvi</sup>	53.22 (4)
Ca1 <sup>xi</sup> —Al1—Ca1 <sup>ii</sup>	131.96 (7)	Pt1—Ca1—Ca1 <sup>xvi</sup>	104.19 (8)
Ca1 <sup>ix</sup> —Al1—Ca1 <sup>ii</sup>	131.96 (7)	All <sup>xiv</sup> —Ca1—Ca1 <sup>xvi</sup>	140.93 (5)
Pt1 <sup>viii</sup> —Al1—Ca1	132.22 (7)	All <sup>iv</sup> —Cal—Cal <sup>xvi</sup>	151.87 (11)
Pt1 <sup>ii</sup> —Al1—Ca1	58.71 (5)	Al1 <sup>xv</sup> —Ca1—Ca1 <sup>xvi</sup>	92.82 (5)
Pt1 <sup>iii</sup> —Al1—Ca1	122.54 (12)	Al1 <sup>ii</sup> —Ca1—Ca1 <sup>xvi</sup>	58.83 (7)
Pt1 <sup>ix</sup> —Al1—Ca1	58.83 (6)	Al1—Ca1—Ca1 <sup>xvi</sup>	104.61 (9)
Ca1 <sup>x</sup> —Al1—Ca1	116.88 (8)	Al1 <sup>xiii</sup> —Ca1—Ca1 <sup>xvi</sup>	57.44 (6)

Symmetry codes: (i) x+1, y, z; (ii) -x+1, -y+1, -z+1; (iii) -x+1, -y, -z+1; (iv) x+1/2, y, -z+1/2; (v) x+1/2, y-1, -z+1/2; (vi) x, y-1, z; (vii) -x+3/2, -y+1, z+1/2; (viii) x-1, y, z; (ix) x-1/2, y, -z+1/2; (x) -x+1/2, -y+1, z+1/2; (xi) x-1/2, y-1, -z+1/2; (xii) x-1/2, y+1, -z+1/2; (xiii) x, y+1, z; (xiv) -x+1/2, -y+1, z-1/2; (xv) x+1/2, y+1, -z+1/2; (xvi) -x+1, -y+2, -z+1.