# inorganic compounds



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# Rietveld refinement of KLaTiO<sub>4</sub> from X-ray powder data

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Key indicators: powder X-ray study; T = 298 K; mean  $\sigma(La-O) = 0.003$  Å; R factor = 0.046; wR factor = 0.068; data-to-parameter ratio = 98.0.

Potassium lanthanum titanate(IV), KLaTiO<sub>4</sub>, has been synthesized by conventional solid-state reaction. It crystallizes isotypically with the NaLnTiO<sub>4</sub> (Ln = La, Pr, Nd, Sm, Eu, Gd, Y and Lu) family. Five of the six atoms in the asymmetric unit (one K, one La, one Ti and two O atoms) are situated on sites with 4mm symmetry, whereas one O atom has 2mm. site symmetry. The crystal structure can be described as being composed of single layers of distorted corner-sharing TiO<sub>6</sub> octahedra extending parallel to (001). The layers are alternately separated by K<sup>+</sup> and La<sup>3+</sup> cations along [001]. The coordination number of both K<sup>+</sup> and La<sup>3+</sup> cations is nine, resulting in distorted KO<sub>9</sub> and LaO<sub>9</sub> polyhedra.

#### **Related literature**

For the isotypic NaLnTiO<sub>4</sub> (Ln = La, Pr, Nd, Sm, Eu, Gd, Y and Lu) family, see: Toda *et al.* (1996a). Orthorhombic symmetry for other members of this family has been reported by Nishimoto *et al.* (2006). Decomposition products of NaLnTiO<sub>4</sub> were investigated by Toda *et al.* (1996b). For preparation by ion-exchange and structure analysis of KLnTiO<sub>4</sub> (Ln = La, Nd, Sm, Eu, Gd, Dy) compounds, see: Schaak & Mallouk (2001). For hydrothermal preparation of similar compounds, see: Dairong *et al.* (1999). For crystallographic background, see: Howard (1982); Thompson *et al.* (1987).

# **Experimental**

Crystal data

KLaTiO<sub>4</sub> a = 3.84155 (10) Å  $M_r = 289.90$  c = 13.4695 (4) ÅTetragonal, P4/nmm  $V = 198.78 (1) \text{ Å}^3$  Z = 2Cu  $K\alpha$  radiation,  $\lambda = 1.54060$ , 1.54443 Å T = 298 Kflat sheet,  $20 \times 20 \text{ mm}$ 

Data collection

 $\begin{array}{ll} \text{PANalytical X'pert PRO} & \text{Data cc} \\ \text{diffractometer} & \text{Scan m} \\ \text{Specimen mounting: packed powder} & 2\theta_{\min} = \\ \text{pellet} & 2\theta_{\text{etan}} \end{array}$ 

Data collection mode: reflection Scan method: continuous  $2\theta_{\min} = 9.872^{\circ}, 2\theta_{\max} = 109.815^{\circ}, 2\theta_{\text{step}} = 0.017^{\circ}$ 

Refinement

 $R_{\rm p} = 0.046$   $\chi^2 = 2.220$   $R_{\rm wp} = 0.068$  5880 data points  $R_{\rm exp} = 0.046$  60 parameters  $R(F^2) = 0.047$ 

Table 1 Selected bond lengths (Å).

K1-O1	3.065 (4)	La1-O3i	2.7628 (12)
K1-O2	2.765 (9)	Ti1-O1	1.9635 (12)
$K1-O2^{i}$	2.7242 (7)	Ti1-O2 <sup>iii</sup>	1.775 (9)
La1-O1	2.530 (3)	Ti1-O3 <sup>iii</sup>	2.558 (7)
La1-O3 <sup>ii</sup>	2.339 (7)		` '

Symmetry codes: (i) -x, -y, -z + 1; (ii) x, y, z + 1; (iii) -x + 1, -y + 1, -z + 1.

Data collection: X'pert Data Collector (PANalytical, 2003); cell refinement: GSAS (Larson & Von Dreele, 2004) and EXPGUI (Toby, 2001); data reduction: X'pert Data Collector; method used to solve structure: coordinates taken from an isotypic compound (Toda et al., 1996b); program(s) used to refine structure: GSAS and EXPGUI; molecular graphics: DIAMOND (Brandenburg, 1999); software used to prepare material for publication: publCIF (Westrip, 2010).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2446).

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# supporting information

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# Rietveld refinement of KLaTiO<sub>4</sub> from X-ray powder data

# **Bai-Chuan Zhu and Kai-Bin Tang**

#### S1. Comment

Layered perovskites that belong to the Ruddlesden-Popper  $A'_2[A_{n-1}B_2O_{3n+1}]$  familiy (A' = alkali, A = alkaline earth or rare earth cation; B = transition metal cation) possess a variety of interesting properties, such as superconductivity, colossal magnetoresistance, ferroelectricity, as well as catalytic activity. The structure of KLaTiO<sub>4</sub> we report here is a n = 1 member of this familiy. Isotypic crystal structures have been reported for NaLnTiO<sub>4</sub> (Ln = La, Pr, Nd, Sm, Eu, Gd, Y and Lu; Toda et al., 1996a) in the space group P4/nmm.

Schaak & Mallouk (2001) reported the  $KLnTiO_4$  (Ln= La, Nd, Sm, Eu, Gd, Dy) family of compounds to crystallize in space group Pbcm, as determined from Rietveld refinements of X-ray powder data. We tested both Pbcm and P4/nmm space groups with the underlying structures  $KLnTiO_4$  (P4/nmm; Schaak & Mallouk, 2001) and  $NaLnTiO_4$  (P4/nmm; Toda  $et\ al.$ , 1996a) as starting models for Rietveld refinement of  $KLaTiO_4$ . The results revealed the P4/nmm model to be significantly better than the Pbcm model. It is well-know that different rare earth elements can affect the crystal structure dramatically. In single layer Ruddlesden-Popper phase perovskites some studies reported that  $NaLnTiO_4$  compounds have tetragonal symmetry for Ln = La—Nd, while an orthorhombic symmetry is observed for Ln = Sm—Lu and Y (Nishimoto  $et\ al.$ , 2006). We can infer that a similar situation might be present for  $KLnTiO_4$  compounds. We ascribe the difference in symmetry between  $KLaTiO_4$  obtained through solid state reactions (tetragonal) and through ion-exchange (orthorhombic) to the different temperature treatment (higher temperatures for the solid state reaction route).

Other methods used to prepare KLaTiO<sub>4</sub> have been reported previously, like an ion exchange method by Schaak & Mallouk (2001) and a hydrothermal method by Dairong *et al.* (1999). To our knowledge, a solid state route to synthesize this compound and its detailed structure analysis based on Rietveld refinement from X-ray powder diffraction data has not been reported. KLaTiO<sub>4</sub> easily decomposes at high temperature and is converted into the three-layer Ruddlesden-Popper phase K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub>. This phenomenon is also found in during preparation of NaLaTiO<sub>4</sub> reported by Toda *et al.* (1996*b*). Therefore we modified the reaction conditions on the basis of the preparation of NaLaTiO<sub>4</sub> and obtained a single phase product successfully.

Fig. 1 shows the observed difference plots (calculated, observed) of the Rietveld refinement.

Fig. 2 illustrates the structure of KLaTiO<sub>4</sub>. It consists of a single layer of corner-sharing distorted TiO<sub>6</sub> octahedra extending parallel to (001). The layers are separated by alternating layers of K<sup>+</sup> and La<sup>3+</sup> cations along [001]. The TiO<sub>6</sub> octahedra (4mm symmetry) are considerably distorted. They have four equal equatorial Ti—O distances [1.9635 (12) Å], one very short Ti—O distance [1.775 (9) Å] toward the K layer and a significantly longer Ti—O distance [2.558 (7) Å] towards the La layer. The corresponding coordination polyhedra around the K<sup>+</sup> and La<sup>3+</sup> cations are distorted KO<sub>9</sub> and LaO<sub>9</sub> polyhedra, each with 4mm symmetry.

### S2. Experimental

The sample was prepared by conventional solid-state reaction. The starting materials were  $KNO_3$ ,  $La_2O_3$  and  $TiO_2$  in a molar ratio of 2:1:2. An excess of  $KNO_3$  (55 mol%) was added to compensate for the loss due to the volatilization of the potassium component.  $La_2O_3$  was heated to 1173 K for 10 h prior to use to remove water and carbonate impurities. The mixture was then ground and calcined at 1223 K for 30 min.

### S3. Refinement

The crystal structure of NaLaTiO<sub>4</sub> (Toda *et al.*, 1996*b*) in the spacegroup *P*4/*nmm* was used as a starting model for the final Rietveld refinement of the KLaTiO<sub>4</sub> structure. Isotropic displacement parameters were used for all atoms. The March-Dollase option in the *EXPGUI* program (Toby, 2001) was applied to correct for preferential orientation along [00*I*] which is often observed for such layered perovskites.

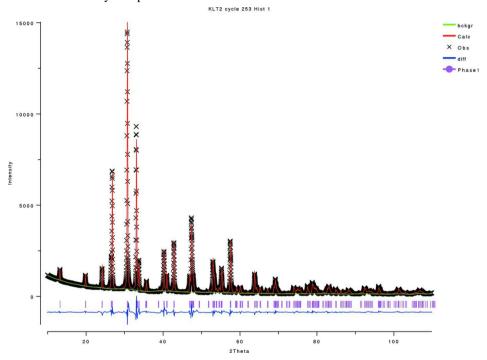


Figure 1
Rietveld difference plot for the refinement of KLaTiO<sub>4</sub>.

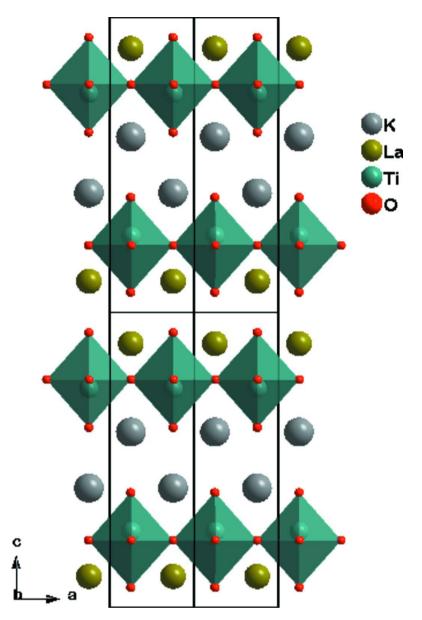


Figure 2
The crystal structure of KLaTiO<sub>4</sub> in a projection along [010].

# Potassium lanthanum titanate

Crystal data KLaTiO<sub>4</sub>  $M_r = 289.90$ Tetragonal, P4/nmmHall symbol: -p 4a 2a a = 3.84155 (10) Å c = 13.4695 (4) Å $V = 198.78 (1) \text{ Å}^3$ 

Z=2  $D_{\rm x}=4.848~{
m Mg~m^{-3}}$  Cu  $K\alpha$  radiation,  $\lambda=1.54060,~1.54443~{
m \AA}$   $T=298~{
m K}$  white flat sheet,  $20\times20~{
m mm}$  Specimen preparation: Prepared at 1223 K

Data collection

PANalytical X'pert PRO diffractometer

Radiation source: sealed tube Graphite monochromator

Refinement

Refinement on  $F^2$ 

Least-squares matrix: full

 $R_{\rm p} = 0.046$ 

 $R_{\rm wp} = 0.068$ 

 $R_{\rm exp} = 0.046$ 

 $R(F^2) = 0.04713$ 

 $\chi^2 = 2.220$ 

5880 data points

Excluded region(s): none

Profile function: CW Profile function number 2 with 18 terms Profile coefficients for Simpson's rule integration of pseudovoigt function C.J. Howard (1982). J. Appl. Cryst.,15,615-620. P. Thompson, D.E. Cox & J.B. Hastings (1987). J. Appl. Cryst.,20,79-83. #1(GU) = 0.000 #2(GV) = -2.261 #3(GW) = -9.290 #4(LX) = 4.310 #5(LY) = 17.630 #6(trns) = 0.000 #7(asym) = 3.5282 #8(shft) = 0.0000 #9(GP) = 17.284 #10(stec) = 0.00 #11(ptec) = 0.00 #12(sfec) = 0.00 #13(L11) = 0.000 #14(L22) = 0.000 #15(L33) = 0.000 #16(L12) = 0.000 #17(L13) = 0.000 #18(L23) = 0.000 Peak tails are ignored where the intensity is below 0.0010 times the peak Aniso. broadening axis 0.0 0.0 1.0

Specimen mounting: packed powder pellet

Data collection mode: reflection

Scan method: continuous

 $2\theta_{\min} = 9.872^{\circ}, 2\theta_{\max} = 109.815^{\circ}, 2\theta_{\text{step}} = 0.017^{\circ}$ 

60 parameters

0 restraints

 $w = 1/[\sigma^2(F_0^2) + (0.0677P)^2]$ 

where  $P = (F_0^2 + 2F_c^2)/3$ 

 $(\Delta/\sigma)_{\rm max} = 0.01$ 

Background function: GSAS Background

function number 1 with 36 terms. Shifted

Chebyshev function of 1st kind 1: 353.285 2:

-361.136 3: 220.846 4: -104.260 5: 61.8271 6:

-33.1030 7: 19.7877 8: -5.01446 9: 3.42337 10:

-3.14370 11: 0.340114 12: 2.15882 13:

3.1 13 7 0 11. 0.3 10111 12. 2.13 002 13.

-0.130836 14: -1.88421 15: 5.08631 16:

-1.48077 17: 4.42719 18: 2.91556 19:

-3.924060E-0220: 0.679453 21: 5.77738 22:

-2.47188 23: 3.81643 24: 3.21357 25: -4.71396

26: -1.63350 27: 0.665874 28: -7.16378 29:

-7.040150E-0230: 3.04932 31: -2.36381 32:

0.787399 33: 4.27144 34: -2.96952 35: 4.90415

36: 1.54599

Preferred orientation correction: March-Dollase AXIS 1 Ratio= 0.96438 h= 0.000 k= 0.000 l=

1.000 Prefered orientation correction range:

Min= 0.94706, Max= 1.11492

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters  $(\mathring{A}^2)$ 

	x	y	Z	$U_{ m iso}$ */ $U_{ m eq}$	
K1	0.25	0.25	0.5950(2)	0.0278 (9)*	
LA1	0.25	0.25	0.89446 (6)	0.0199 (4)*	
TI1	0.75	0.75	0.74203 (19)	0.0151 (7)*	
O1	0.75	0.25	0.7723 (4)	0.0189 (17)*	
O2	0.25	0.25	0.3897 (6)	0.039 (2)*	
O3	0.25	0.25	0.0681 (5)	0.018 (2)*	

### Geometric parameters (Å, °)

K1—O1 <sup>i</sup>	3.065 (4)	La1—O1 <sup>iii</sup>	2.530(3)
K1—01	3.065 (4)	La1—O3viii	2.339 (7)
K1—O1 <sup>ii</sup>	3.065 (4)	La1—O3iv	2.7628 (12)
K1—O1 <sup>iii</sup>	3.065 (4)	La1—O3 <sup>v</sup>	2.7628 (12)
K1—O2	2.765 (9)	La1—O3 <sup>vi</sup>	2.7628 (12)
K1—O2iv	2.7242 (7)	La1—O3 <sup>vii</sup>	2.7628 (12)
K1—O2 <sup>v</sup>	2.7242 (7)	Ti1—O1	1.9635 (12)
K1—O2 <sup>vi</sup>	2.7242 (7)	Ti1—O1ix	1.9635 (12)

# supporting information

V1 O2vii	2 7242 (7)	T:1 O1iii	1.0625 (12)
K1—02 <sup>vii</sup>	2.7242 (7)	Ti1—01 <sup>iii</sup>	1.9635 (12)
La1—O1 <sup>i</sup> La1—O1	2.530 (3)	Ti1—O1 <sup>x</sup> Ti1—O2 <sup>vii</sup>	1.9635 (12)
La1—O1	2.530 (3)	Ti1—03 <sup>vii</sup>	1.775 (9)
Lai—Oi"	2.530 (3)	111—03	2.558 (7)
O1 <sup>i</sup> —K1—O1	77.62 (13)	O1 <sup>i</sup> —La1—O3 <sup>v</sup>	65.85 (12)
O1 <sup>i</sup> —K1—O1 <sup>ii</sup>	52.61 (8)	O1 <sup>i</sup> —La1—O3 <sup>vi</sup>	130.30 (11)
O1 <sup>i</sup> —K1—O1 <sup>iii</sup>	52.61 (8)	O1 <sup>i</sup> —La1—O3 <sup>vii</sup>	130.30 (11)
O1 <sup>i</sup> —K1—O2	141.19 (6)	O1—La1—O1 <sup>ii</sup>	64.95 (9)
$O1^{i}$ — $K1$ — $O2^{iv}$	59.94 (15)	O1—La1—O1 <sup>iii</sup>	64.95 (9)
O1 <sup>i</sup> —K1—O2 <sup>v</sup>	59.94 (15)	O1—La1—O3viii	130.59 (9)
$O1^{i}$ — $K1$ — $O2^{vi}$	112.52 (18)	O1—La1—O3 <sup>iv</sup>	130.30 (11)
O1 <sup>i</sup> —K1—O2 <sup>vii</sup>	112.52 (18)	O1—La1—O3 <sup>v</sup>	130.30 (11)
O1—K1—O1 <sup>ii</sup>	52.61 (8)	O1—La1—O3 <sup>vi</sup>	65.85 (12)
O1—K1—O1 <sup>iii</sup>	52.61 (8)	O1—La1—O3 <sup>vii</sup>	65.85 (12)
O1—K1—O2	141.19 (6)	O1 <sup>ii</sup> —La1—O1 <sup>iii</sup>	98.81 (17)
O1—K1—O2 <sup>iv</sup>	112.52 (18)	O1 <sup>ii</sup> —La1—O3 <sup>viii</sup>	130.59 (9)
O1—K1—O2 <sup>v</sup>	112.52 (18)	O1 <sup>ii</sup> —La1—O3 <sup>iv</sup>	65.85 (12)
O1—K1—O2 <sup>vi</sup>	59.94 (15)	O1 <sup>ii</sup> —La1—O3 <sup>v</sup>	130.30 (11)
O1—K1—O2 <sup>vii</sup>	59.94 (15)	O1 <sup>ii</sup> —La1—O3 <sup>vi</sup>	65.85 (12)
O1"—K1—O1"	77.62 (13)	O1 <sup>ii</sup> —La1—O3 <sup>vii</sup>	130.30 (11)
O1 <sup>ii</sup> —K1—O2	141.19 (6)	O1 <sup>iii</sup> —La1—O3 <sup>viii</sup>	130.59 (9)
O1 <sup>ii</sup> —K1—O2 <sup>iv</sup>	59.94 (15)	O1 <sup>iii</sup> —La1—O3 <sup>iv</sup>	130.30 (11)
O1 <sup>ii</sup> —K1—O2 <sup>v</sup>	112.52 (18)	O1 <sup>iii</sup> —La1—O3 <sup>v</sup>	65.85 (12)
$O1^{ii}$ — $K1$ — $O2^{vi}$	59.94 (15)	O1 <sup>iii</sup> —La1—O3 <sup>vi</sup>	130.30 (11)
$O1^{ii}$ — $K1$ — $O2^{vii}$	112.52 (18)	O1 <sup>iii</sup> —La1—O3 <sup>vii</sup>	65.85 (12)
O1 <sup>iii</sup> —K1—O2	141.19 (6)	O3 <sup>viii</sup> —La1—O3 <sup>iv</sup>	79.48 (14)
$O1^{iii}$ — $K1$ — $O2^{iv}$	112.52 (18)	O3 <sup>viii</sup> —La1—O3 <sup>v</sup>	79.48 (14)
O1 <sup>iii</sup> —K1—O2 <sup>v</sup>	59.94 (15)	O3 <sup>viii</sup> —La1—O3 <sup>vi</sup>	79.48 (14)
O1 <sup>iii</sup> —K1—O2 <sup>vi</sup>	112.52 (18)	O3 <sup>viii</sup> —La1—O3 <sup>vii</sup>	79.48 (14)
O1 <sup>iii</sup> —K1—O2 <sup>vii</sup>	59.94 (15)	O3 <sup>iv</sup> —La1—O3 <sup>v</sup>	88.09 (5)
O2—K1—O2 <sup>iv</sup>	94.34 (18)	O3 <sup>iv</sup> —La1—O3 <sup>vi</sup>	88.09 (5)
O2—K1—O2 <sup>v</sup>	94.34 (18)	O3 <sup>iv</sup> —La1—O3 <sup>vii</sup>	159.0 (3)
O2—K1—O2 <sup>vi</sup>	94.34 (18)	O3 <sup>v</sup> —La1—O3 <sup>vi</sup>	159.0 (3)
O2—K1—O2 <sup>vii</sup>	94.34 (18)	O3 <sup>v</sup> —La1—O3 <sup>vii</sup>	88.09 (5)
$O2^{iv}$ — $K1$ — $O2^{v}$	89.67 (3)	O3 <sup>vi</sup> —La1—O3 <sup>vii</sup>	88.09 (5)
$O2^{iv}$ — $K1$ — $O2^{vi}$	89.67 (3)	O1—Ti1—O1 <sup>ix</sup>	156.1 (3)
$O2^{iv}$ — $K1$ — $O2^{vii}$	171.3 (4)	O1—Ti1—O1 <sup>iii</sup>	87.53 (7)
$O2^{v}$ — $K1$ — $O2^{vi}$	171.3 (4)	O1—Ti1—O1 <sup>x</sup>	87.53 (7)
O2 <sup>v</sup> —K1—O2 <sup>vii</sup>	89.67 (3)	O1—Ti1—O2 <sup>vii</sup>	101.97 (16)
$O2^{vi}$ — $K1$ — $O2^{vii}$	89.67 (3)	O1 <sup>ix</sup> —Ti1—O1 <sup>iii</sup>	87.53 (7)
O1 <sup>i</sup> —La1—O1	98.81 (17)	$O1^{ix}$ — $Ti1$ — $O1^{x}$	87.53 (7)
O1 <sup>i</sup> —La1—O1 <sup>ii</sup>	64.95 (9)	$O1^{ix}$ — $Ti1$ — $O2^{vii}$	101.97 (16)
O1 <sup>i</sup> —La1—O1 <sup>iii</sup>	64.95 (9)	O1 <sup>iii</sup> —Ti1—O1 <sup>x</sup>	156.1 (3)

# supporting information

O1 <sup>i</sup> —La1—O3 <sup>viii</sup>	130.59 (9)	O1 <sup>iii</sup> —Ti1—O2 <sup>vii</sup>	101.97 (16)
O1 <sup>i</sup> —La1—O3 <sup>iv</sup>	65.85 (12)	O1×—Ti1—O2 <sup>vii</sup>	101.97 (16)

Symmetry codes: (i) x-1, y, z; (ii) -y+1/2, x-1, z; (iii) -y+1/2, x, z; (iv) -x, -y, -z+1; (v) -x, -y+1, -z+1; (vi) -x+1, -y, -z+1; (vii) -x+1, -y+1, -z+1; (viii) x, y, z+1; (ix) x, y+1, z; (x) -y+3/2, x, z.