## Structure Reports

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# Rietveld refinement of $\mathrm{KLaTiO}_{4}$ from X-ray powder data 

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Key indicators: powder X-ray study; $T=298 \mathrm{~K}$; mean $\sigma(\mathrm{La-O})=0.003 \AA ; R$ factor $=$ $0.046 ; w R$ factor $=0.068$; data-to-parameter ratio $=98.0$.

Potassium lanthanum titanate(IV), $\mathrm{KLaTiO}_{4}$, has been synthesized by conventional solid-state reaction. It crystallizes isotypically with the $\mathrm{Na} L n \mathrm{TiO}_{4}(L n=\mathrm{La}, \mathrm{Pr}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Eu}, \mathrm{Gd}$, Y and Lu ) family. Five of the six atoms in the asymmetric unit (one K , one La , one Ti and two O atoms) are situated on sites with 4 mm symmetry, whereas one O atom has 2 mm . site symmetry. The crystal structure can be described as being composed of single layers of distorted corner-sharing $\mathrm{TiO}_{6}$ octahedra extending parallel to (001). The layers are alternately separated by $\mathrm{K}^{+}$and $\mathrm{La}^{3+}$ cations along [001]. The coordination number of both $\mathrm{K}^{+}$and $\mathrm{La}^{3+}$ cations is nine, resulting in distorted $\mathrm{KO}_{9}$ and $\mathrm{LaO}_{9}$ polyhedra.

## Related literature

For the isotypic $\mathrm{Na}_{2} \mathrm{LnTiO}_{4}$ ( $L n=\mathrm{La}, \mathrm{Pr}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Eu}, \mathrm{Gd}, \mathrm{Y}$ and Lu) family, see: Toda et al. (1996a). Orthorhombic symmetry for other members of this family has been reported by Nishimoto et al. (2006). Decomposition products of NaLnTiO 4 were investigated by Toda et al. (1996b). For preparation by ion-exchange and structure analysis of $K L n \mathrm{TiO}_{4}$ ( $L n=\mathrm{La}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Eu}, \mathrm{Gd}, \mathrm{Dy}$ ) compounds, see: Schaak \& Mallouk (2001). For hydrothermal preparation of similar compounds, see: Dairong et al. (1999). For crystallographic background, see: Howard (1982); Thompson et al. (1987).

## Experimental

## Crystal data

$\mathrm{KLaTiO}_{4}$
$M_{r}=289.90$
Tetragonal, $P 4 / \mathrm{nmm}$

$$
\begin{aligned}
& a=3.84155(10) \AA \\
& c=13.4695(4) \AA \\
& V=198.78(1) \AA^{3}
\end{aligned}
$$

$Z=2$
$\mathrm{Cu} \mathrm{K} \mathrm{\alpha}$ radiation, $\lambda=1.54060$,
$1.54443 \AA$
collection
PANalytical X'pert PRO diffractometer
Specimen mounting: packed powder pellet

## Refinement

$R_{\mathrm{p}}=0.046$
$\chi^{2}=2.220$
$R_{\text {wp }}=0.068$
5880 data points
$R_{\text {exp }}=0.046$
60 parameters

Table 1
Selected bond lengths ( $\AA$ ).

| $\mathrm{K} 1-\mathrm{O} 1$ | $3.065(4)$ | $\mathrm{La} 1-\mathrm{O}^{\mathrm{i}}$ | $2.7628(12)$ |
| :--- | :--- | :--- | :--- |
| $\mathrm{K} 1-\mathrm{O} 2$ | $2.765(9)$ | $\mathrm{Ti} 1-\mathrm{O} 1$ | $1.9635(12)$ |
| $\mathrm{K} 1-\mathrm{O} 2^{\mathrm{i}}$ | $2.7242(7)$ | Ti1-O2 $2^{\mathrm{iii}}$ | $1.775(9)$ |
| $\mathrm{La} 1-\mathrm{O} 1$ | $2.530(3)$ | Ti1-O3 $^{\mathrm{iii}}$ | $2.558(7)$ |
| $\mathrm{La} 1-\mathrm{O} 3^{\mathrm{ii}}$ | $2.339(7)$ |  |  |

Symmetry codes: (i) $-x,-y,-z+1$; (ii) $x, y, z+1$; (iii) $-x+1,-y+1,-z+1$.
Data collection: X'pert Data Collector (PANalytical, 2003); cell refinement: GSAS (Larson \& Von Dreele, 2004) and EXPGUI (Toby, 2001); data reduction: X'pert Data Collector; method used to solve structure: coordinates taken from an isotypic compound (Toda et al., 1996b); program(s) used to refine structure: GSAS and EXPGUI; molecular graphics: DIAMOND (Brandenburg, 1999); software used to prepare material for publication: publCIF (Westrip, 2010).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2446).

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## supporting information

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## Rietveld refinement of $\mathrm{KLaTiO}_{4}$ from X-ray powder data

## Bai-Chuan Zhu and Kai-Bin Tang

## S1. Comment

Layered perovskites that belong to the Ruddlesden-Popper $A^{\prime}\left[A_{n-1} B_{2} \mathrm{O}_{3 n+1}\right]$ familiy ( $A^{\prime}=$ alkali, $A=$ alkaline earth or rare earth cation; $B=$ transition metal cation) possess a variety of interesting properties, such as superconductivity, colossal magnetoresistance, ferroelectricity, as well as catalytic activity. The structure of $\mathrm{KLaTiO}_{4}$ we report here is a $n=1$ member of this familiy. Isotypic crystal structures have been reported for $\mathrm{Na} L n \mathrm{TiO}_{4}(\operatorname{Ln}=\mathrm{La}, \mathrm{Pr}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Eu}, \mathrm{Gd}, \mathrm{Y}$ and Lu ; Toda et al., 1996a) in the space group P4/nmm.
Schaak \& Mallouk (2001) reported the $\mathrm{KLnTiO}_{4}(\operatorname{Ln}=\mathrm{La}, \mathrm{Nd}, \mathrm{Sm}, \mathrm{Eu}, \mathrm{Gd}, \mathrm{Dy})$ family of compounds to crystallize in space group Pbcm , as determined from Rietveld refinements of X-ray powder data. We tested both Pbcm and $P 4 / \mathrm{nmm}$ space groups with the underlying structures $\mathrm{KLnTiO}_{4}(P 4 / n m m$; Schaak \& Mallouk, 2001) and $\mathrm{NaLnTiO} 4(P 4 / n m m$; Toda et al., 1996a) as starting models for Rietveld refinement of $\mathrm{KLaTiO}_{4}$. The results revealed the $P 4 / \mathrm{nmm}$ model to be significantly better than the Pbcm model. It is well-know that different rare earth elements can affect the crystal structure dramatically. In single layer Ruddlesden-Popper phase perovskites some studies reported that $\mathrm{Na} L n \mathrm{TiO}_{4}$ compounds have tetragonal symmetry for $L n=\mathrm{La}-\mathrm{Nd}$, while an orthorhombic symmetry is observed for $L n=\mathrm{Sm}-\mathrm{Lu}$ and Y (Nishimoto et al., 2006). We can infer that a similar situation might be present for $\mathrm{K}_{\mathrm{Ln}} \mathrm{TiO}_{4}$ compounds. We ascribe the difference in symmetry between $\mathrm{KLaTiO}_{4}$ obtained through solid state reactions (tetragonal) and through ion-exchange (orthorhombic) to the different temperature treatment (higher temperatures for the solid state reaction route).
Other methods used to prepare $\mathrm{KLaTiO}_{4}$ have been reported previously, like an ion exchange method by Schaak \& Mallouk (2001) and a hydrothermal method by Dairong et al. (1999). To our knowledge, a solid state route to synthesize this compound and its detailed structure analysis based on Rietveld refinement from X-ray powder diffraction data has not been reported. $\mathrm{KLaTiO}_{4}$ easily decomposes at high temperature and is converted into the three-layer RuddlesdenPopper phase $\mathrm{K}_{2} \mathrm{La}_{2} \mathrm{Ti}_{3} \mathrm{O}_{10}$. This phenomenon is also found in during preparation of $\mathrm{NaLaTiO}_{4}$ reported by Toda et al. (1996b). Therefore we modified the reaction conditions on the basis of the preparation of $\mathrm{NaLaTiO}_{4}$ and obtained a single phase product successfully.

Fig. 1 shows the observed difference plots (calculated, observed) of the Rietveld refinement.
Fig. 2 illustrates the structure of $\mathrm{KLaTiO}_{4}$. It consists of a single layer of corner-sharing distorted $\mathrm{TiO}_{6}$ octahedra extending parallel to (001). The layers are separated by alternating layers of $\mathrm{K}^{+}$and $\mathrm{La}^{3+}$ cations along [001]. The $\mathrm{TiO}_{6}$ octahedra ( 4 mm symmetry) are considerably distorted. They have four equal equatorial $\mathrm{Ti}-\mathrm{O}$ distances [1.9635 (12) $\AA$ ], one very short $\mathrm{Ti}-\mathrm{O}$ distance $[1.775(9) \AA]$ toward the K layer and a significantly longer $\mathrm{Ti}-\mathrm{O}$ distance $[2.558$ (7) $\AA$ ] towards the La layer. The corresponding coordination polyhedra around the $\mathrm{K}^{+}$and $\mathrm{La}^{3+}$ cations are distorted $\mathrm{KO}_{9}$ and $\mathrm{LaO}_{9}$ polyhedra, each with 4 mm symmetry.

## S2. Experimental

The sample was prepared by conventional solid-state reaction. The starting materials were $\mathrm{KNO}_{3}, \mathrm{La}_{2} \mathrm{O}_{3}$ and $\mathrm{TiO}_{2}$ in a molar ratio of 2:1:2. An excess of $\mathrm{KNO}_{3}(55 \mathrm{~mol} \%)$ was added to compensate for the loss due to the volatilization of the potassium component. $\mathrm{La}_{2} \mathrm{O}_{3}$ was heated to 1173 K for 10 h prior to use to remove water and carbonate impurities. The mixture was then ground and calcined at 1223 K for 30 min .

## S3. Refinement

The crystal structure of $\mathrm{NaLaTiO}_{4}$ (Toda et al., 1996b) in the spacegroup $P 4 / n m m$ was used as a starting model for the final Rietveld refinement of the $\mathrm{KLaTiO}_{4}$ structure. Isotropic displacement parameters were used for all atoms. The March-Dollase option in the EXPGUI program (Toby, 2001) was applied to correct for preferential orientation along [00l] which is often observed for such layered perovskites.


## Figure 1

Rietveld difference plot for the refinement of $\mathrm{KLaTiO}_{4}$.


Figure 2
The crystal structure of $\mathrm{KLaTiO}_{4}$ in a projection along [010].

## Potassium lanthanum titanate

Crystal data
$\mathrm{KLaTiO}_{4}$
$M_{r}=289.90$
Tetragonal, $P 4 / \mathrm{nmm}$
Hall symbol: -p 4a 2a
$a=3.84155$ (10) $\AA$
$c=13.4695(4) \AA$
$V=198.78(1) \AA^{3}$
$Z=2$
$D_{\mathrm{x}}=4.848 \mathrm{Mg} \mathrm{m}^{-3}$
$\mathrm{Cu} K \alpha$ radiation, $\lambda=1.54060,1.54443 \AA$
$T=298 \mathrm{~K}$
white
flat sheet, $20 \times 20 \mathrm{~mm}$
Specimen preparation: Prepared at 1223 K

## Data collection

PANalytical X'pert PRO
diffractometer
Radiation source: sealed tube
Graphite monochromator

## Refinement

Refinement on $F^{2}$
Least-squares matrix: full
$R_{\mathrm{p}}=0.046$
$R_{\text {wp }}=0.068$
$R_{\text {exp }}=0.046$
$R\left(F^{2}\right)=0.04713$
$\chi^{2}=2.220$
5880 data points
Excluded region(s): none
Profile function: CW Profile function number 2 with 18 terms Profile coefficients for Simpson's rule integration of pseudovoigt function C.J. Howard (1982). J. Appl. Cryst., 15,615-620. P. Thompson, D.E. Cox \& J.B. Hastings (1987). J. Appl. Cryst.,20,79-83. \#1(GU) $=0.000$ \#2(GV) $=-2.261 \# 3(\mathrm{GW})=-9.290 \# 4(\mathrm{LX})=4.310$ $\# 5(\mathrm{LY})=17.630 \# 6($ trns $)=0.000 \# 7($ asym $)=$ $3.5282 \# 8($ shft $)=0.0000 \# 9(\mathrm{GP})=17.284$ $\# 10(\mathrm{stec})=0.00 \# 11(\mathrm{ptec})=0.00 \# 12(\mathrm{sfec})=0.00$ $\# 13(\mathrm{~L} 11)=0.000 \# 14(\mathrm{~L} 22)=0.000 \# 15(\mathrm{~L} 33)=$ $0.000 \# 16(\mathrm{~L} 12)=0.000 \# 17(\mathrm{~L} 13)=0.000$ \#18(L23) $=0.000$ Peak tails are ignored where the intensity is below 0.0010 times the peak Aniso. broadening axis 0.00 .01 .0

Specimen mounting: packed powder pellet
Data collection mode: reflection
Scan method: continuous
$2 \theta_{\min }=9.872^{\circ}, 2 \theta_{\max }=109.815^{\circ}, 2 \theta_{\text {step }}=0.017^{\circ}$

60 parameters
0 restraints
$w=1 /\left[\sigma^{2}\left(F_{\mathrm{o}}{ }^{2}\right)+(0.0677 P)^{2}\right]$
where $P=\left(F_{\mathrm{o}}{ }^{2}+2 F_{\mathrm{c}}{ }^{2}\right) / 3$
$(\Delta / \sigma)_{\max }=0.01$
Background function: GSAS Background
function number 1 with 36 terms. Shifted
Chebyshev function of 1st kind 1:353.285 2:
-361.136 3: 220.846 4: -104.260 5: 61.8271 6:
-33.1030 7: 19.7877 8: -5.01446 9: 3.42337 10:
-3.14370 11: 0.340114 12: 2.15882 13:
-0.130836 14: -1.88421 15: 5.0863116 :
-1.48077 17: 4.42719 18: 2.91556 19:
-3.924060E-0220: 0.679453 21: 5.77738 22:
-2.47188 23: 3.81643 24: 3.21357 25: -4.71396
26: -1.63350 27: 0.665874 28: -7.16378 29:
-7.040150E-0230: 3.04932 31: -2.3638132 :
0.787399 33: 4.27144 34: -2.96952 35: 4.90415

36: 1.54599
Preferred orientation correction: March-Dollase
AXIS 1 Ratio $=0.96438 \mathrm{~h}=0.000 \mathrm{k}=0.000 \mathrm{l}=$ 1.000 Prefered orientation correction range:
$\operatorname{Min}=0.94706, \operatorname{Max}=1.11492$

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters ( $\dot{A}^{2}$ )

|  | $x$ | $y$ | $z$ | $U_{\text {iso }}{ }^{*} / U_{\text {eq }}$ |
| :--- | :--- | :--- | :--- | :--- |
| K1 | 0.25 | 0.25 | $0.5950(2)$ | $0.0278(9)^{*}$ |
| LA1 | 0.25 | 0.25 | $0.89446(6)$ | $0.0199(4)^{*}$ |
| TI1 | 0.75 | 0.75 | $0.74203(19)$ | $0.0151(7)^{*}$ |
| O1 | 0.75 | 0.25 | $0.7723(4)$ | $0.0189(17)^{*}$ |
| O2 | 0.25 | 0.25 | $0.3897(6)$ | $0.039(2)^{*}$ |
| O3 | 0.25 | 0.25 | $0.0681(5)$ | $0.018(2)^{*}$ |

Geometric parameters ( $A,{ }^{\circ}$ )

| $\mathrm{K} 1-\mathrm{O} 1^{\mathrm{i}}$ | 3.065 (4) | $\mathrm{La} 1-\mathrm{O} 1^{\text {iii }}$ | 2.530 (3) |
| :---: | :---: | :---: | :---: |
| $\mathrm{K} 1-\mathrm{O} 1$ | 3.065 (4) | La1-O3 ${ }^{\text {viii }}$ | 2.339 (7) |
| $\mathrm{K} 1-\mathrm{O} 1^{\text {ii }}$ | 3.065 (4) | La1-O3 ${ }^{\text {iv }}$ | 2.7628 (12) |
| $\mathrm{K} 1-\mathrm{O} 1^{\text {iii }}$ | 3.065 (4) | $\mathrm{La} 1-\mathrm{O}^{\text {v }}$ | 2.7628 (12) |
| $\mathrm{K} 1-\mathrm{O} 2$ | 2.765 (9) | La1-O3 ${ }^{\text {vi }}$ | 2.7628 (12) |
| $\mathrm{K} 1-\mathrm{O} 2^{\text {iv }}$ | 2.7242 (7) | La1-O3 ${ }^{\text {vii }}$ | 2.7628 (12) |
| $\mathrm{K} 1-\mathrm{O} 2^{\text {v }}$ | 2.7242 (7) | Til-O1 | 1.9635 (12) |
| $\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vi }}$ | 2.7242 (7) | Ti1-O1 ${ }^{\text {ix }}$ | 1.9635 (12) |


| $\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vii }}$ | 2.7242 (7) |
| :---: | :---: |
| La1-O1 ${ }^{\text {i }}$ | 2.530 (3) |
| Lal-O1 | 2.530 (3) |
| $\mathrm{La} 1-\mathrm{O} 1^{\text {ii }}$ | 2.530 (3) |
| O1- ${ }^{\text {i }}$ K1-O1 | 77.62 (13) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 1^{\text {ii }}$ | 52.61 (8) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 1^{\text {iii }}$ | 52.61 (8) |
| $\mathrm{O} 1{ }^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 2$ | 141.19 (6) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 2^{\mathrm{iv}}$ | 59.94 (15) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 2^{\text {v }}$ | 59.94 (15) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 2^{\text {vi }}$ | 112.52 (18) |
| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 112.52 (18) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 1^{\text {ii }}$ | 52.61 (8) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 1^{\text {iii }}$ | 52.61 (8) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 2$ | 141.19 (6) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {iv }}$ | 112.52 (18) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 2^{\text {v }}$ | 112.52 (18) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 2^{\text {vi }}$ | 59.94 (15) |
| $\mathrm{O} 1-\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 59.94 (15) |
| $\mathrm{O} 1^{\text {ii }}-\mathrm{K} 1-\mathrm{O} 1^{\text {iii }}$ | 77.62 (13) |
| $\mathrm{O} 1{ }^{\text {iii }} \mathrm{K} 1-\mathrm{O} 2$ | 141.19 (6) |
| $\mathrm{O} 1^{\mathrm{ii}}-\mathrm{K} 1-\mathrm{O} 2^{\text {iv }}$ | 59.94 (15) |
| $\mathrm{O} 1^{\mathrm{ii}}-\mathrm{K} 1-\mathrm{O} 2^{\mathrm{v}}$ | 112.52 (18) |
| $\mathrm{O} 1^{\mathrm{ii}}-\mathrm{K} 1-\mathrm{O} 2^{\text {vi }}$ | 59.94 (15) |
| $\mathrm{O} 1^{1 i}-\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 112.52 (18) |
| $\mathrm{O} 1^{\text {iii- }} \mathrm{K} 1-\mathrm{O} 2$ | 141.19 (6) |
| $\mathrm{O} 1^{\text {iii }}-\mathrm{K} 1-\mathrm{O} 2^{\text {iv }}$ | 112.52 (18) |
| $\mathrm{O} 1^{\mathrm{iii}}-\mathrm{K} 1-\mathrm{O} 2^{\mathrm{v}}$ | 59.94 (15) |
| $\mathrm{O} 1^{\text {iii }}-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vi }}$ | 112.52 (18) |
| $\mathrm{O}{ }^{\text {iii] }}$ - $\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 59.94 (15) |
| $\mathrm{O} 2-\mathrm{K} 1-\mathrm{O} 2^{\mathrm{iv}}$ | 94.34 (18) |
| $\mathrm{O} 2-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {v }}$ | 94.34 (18) |
| $\mathrm{O} 2-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vi }}$ | 94.34 (18) |
| $\mathrm{O} 2-\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 94.34 (18) |
| $\mathrm{O} 2^{\mathrm{iv}}-\mathrm{K} 1-\mathrm{O} 2^{\mathrm{v}}$ | 89.67 (3) |
| $\mathrm{O} 2^{\text {iv }}-\mathrm{K} 1-\mathrm{O} 2^{\text {vi }}$ | 89.67 (3) |
| $\mathrm{O} 2{ }^{\text {iv }}-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vii }}$ | 171.3 (4) |
| $\mathrm{O} 2{ }^{\mathrm{v}}-\mathrm{K} 1-\mathrm{O} 2^{\text {vi }}$ | 171.3 (4) |
| $\mathrm{O} 2{ }^{\mathrm{v}}-\mathrm{K} 1-\mathrm{O} 2^{\text {vii }}$ | 89.67 (3) |
| $\mathrm{O} 2{ }^{\text {vi }}-\mathrm{K} 1-\mathrm{O} 2{ }^{\text {vii }}$ | 89.67 (3) |
| O1-LLal-O1 | 98.81 (17) |
| $\mathrm{O} 1^{\mathrm{i}}$ - $\mathrm{La} 1-\mathrm{O} 1^{\text {ii }}$ | 64.95 (9) |
| $\mathrm{O} 1{ }^{\text {i }}$ - $\mathrm{La} 1-\mathrm{Ol}^{\text {iii }}$ | 64.95 (9) |


| Ti1-O1 $1^{\text {iii }}$ | 1.9635 (12) |
| :---: | :---: |
| Til-O1 ${ }^{\text {x }}$ | 1.9635 (12) |
| Ti1-O2 ${ }^{\text {vii }}$ | 1.775 (9) |
| Ti1-O3 ${ }^{\text {vii }}$ | 2.558 (7) |
| $\mathrm{O} 1^{\mathrm{i}}$ - $\mathrm{La} 1-\mathrm{O}^{\text {v }}$ | 65.85 (12) |
| O1-La1-O3 ${ }^{\text {vi }}$ | 130.30 (11) |
| $\mathrm{O} 1^{\text {i }}$ - $\mathrm{La} 1-\mathrm{O} 3^{\text {vii }}$ | 130.30 (11) |
| $\mathrm{O} 1-\mathrm{La} 1-\mathrm{O}^{\text {ii }}$ | 64.95 (9) |
| $\mathrm{O} 1-\mathrm{La} 1-\mathrm{O} 1^{\text {iii }}$ | 64.95 (9) |
| O1-La1-O3 ${ }^{\text {viii }}$ | 130.59 (9) |
| $\mathrm{O} 1-\mathrm{La} 1-\mathrm{O}^{\text {iv }}$ | 130.30 (11) |
| O1-La1-O3 ${ }^{\text {v }}$ | 130.30 (11) |
| $\mathrm{O} 1-\mathrm{La} 1-\mathrm{O} 3{ }^{\text {vi }}$ | 65.85 (12) |
| $\mathrm{O} 1-\mathrm{La} 1-3^{\text {vii }}$ | 65.85 (12) |
| $\mathrm{O} 1^{\text {ii- }} \mathrm{La} 1-\mathrm{O} 1^{\text {iii }}$ | 98.81 (17) |
| $\mathrm{O} 1{ }^{\text {ii }}-\mathrm{La} 1-\mathrm{O} 3^{\text {viii }}$ | 130.59 (9) |
| $\mathrm{O} 1^{\text {ii- }} \mathrm{La} 1-\mathrm{O}^{\text {iv }}$ | 65.85 (12) |
| $\mathrm{O} 1^{\mathrm{ii}}-\mathrm{La} 1-\mathrm{O}^{\mathrm{v}}$ | 130.30 (11) |
| $\mathrm{O} 1^{\text {ii- }} \mathrm{La} 1-\mathrm{O}^{\text {vi }}$ | 65.85 (12) |
| $\mathrm{O} 1^{\text {iii- }} \mathrm{La} 1-\mathrm{O}^{\text {vii }}$ | 130.30 (11) |
| $\mathrm{O} 1^{\text {iii }}-\mathrm{La} 1-\mathrm{O} 3^{\text {viii }}$ | 130.59 (9) |
| $\mathrm{O} 1^{\text {iii] }} \mathrm{La} 1-\mathrm{O} 3{ }^{\text {iv }}$ | 130.30 (11) |
| $\mathrm{O} 1^{\text {iii }}$-La1-O3 ${ }^{\text {v }}$ | 65.85 (12) |
| $\mathrm{O} 1^{\text {iiii }} \mathrm{La} 1-\mathrm{O} 3^{\text {vi }}$ | 130.30 (11) |
| $\mathrm{O} 1^{\text {iii }}$-La1-O3 ${ }^{\text {vii }}$ | 65.85 (12) |
| O3 ${ }^{\text {viii }}-\mathrm{La} 1-\mathrm{O} 3^{\text {iv }}$ | 79.48 (14) |
| O3 ${ }^{\text {viii }}$ - $\mathrm{La} 1-3^{\text {v }}$ | 79.48 (14) |
| O 3 viii $-\mathrm{La} 1-\mathrm{O} 3^{\text {vi }}$ | 79.48 (14) |
| O3 ${ }^{\text {viii }}$-La1-O3 $3^{\text {vii }}$ | 79.48 (14) |
| $\mathrm{O}^{\text {iv }}-\mathrm{La} 1-\mathrm{O} 3^{\text {v }}$ | 88.09 (5) |
| $\mathrm{O}^{\text {iv }}-\mathrm{La} 1-\mathrm{O}^{\text {vi }}$ | 88.09 (5) |
| O3 ${ }^{\text {iv }}-\mathrm{La} 1-\mathrm{O} 3{ }^{\text {vii }}$ | 159.0 (3) |
| $\mathrm{O}^{\mathrm{v}}$ - $\mathrm{La} 1-\mathrm{O}^{\text {vi }}$ | 159.0 (3) |
| $\mathrm{O} 3^{v}-\mathrm{La} 1-\mathrm{O} 3^{\text {vii }}$ | 88.09 (5) |
| $\mathrm{O} 3{ }^{\text {vi }}-\mathrm{La} 1-\mathrm{O} 3^{\text {vii }}$ | 88.09 (5) |
| $\mathrm{O} 1-\mathrm{Til}-\mathrm{O} 1^{\text {ix }}$ | 156.1 (3) |
| O1-Til-O1 ${ }^{\text {iii }}$ | 87.53 (7) |
| $\mathrm{O} 1-\mathrm{Ti}-\mathrm{Ol}^{\text {x }}$ | 87.53 (7) |
| $\mathrm{O} 1-\mathrm{Ti} 1-\mathrm{O} 2^{\text {vii }}$ | 101.97 (16) |
| $\mathrm{O} 1^{\mathrm{ix}}-\mathrm{Ti} 1-\mathrm{O}{ }^{1 i i}$ | 87.53 (7) |
| $\mathrm{O} 1^{\mathrm{ix}}$ - Ti1- $\mathrm{O}^{\text {x }}$ | 87.53 (7) |
| $\mathrm{O} 1^{\mathrm{ix}}-\mathrm{Til}-\mathrm{O} 2^{\text {vii }}$ | 101.97 (16) |
| $\mathrm{O} 1^{\text {iii }}-\mathrm{Ti} 1-\mathrm{O} 1^{\mathrm{x}}$ | 156.1 (3) |

## supporting information

| $\mathrm{O} 1^{\mathrm{i}}-\mathrm{La} 1-\mathrm{O} 3^{\text {viii }}$ | 130.59 (9) | $\mathrm{O} 1^{\text {iiii }}$ - $\mathrm{Til}-\mathrm{O} 2^{\text {vii }}$ | 101.97 (16) |
| :---: | :---: | :---: | :---: |
| $\mathrm{O} 1{ }^{\text {i }} \mathrm{La} 1-\mathrm{O}^{\text {iv }}$ | 65.85 (12) | $\mathrm{O} 1^{\mathrm{x}}-\mathrm{Ti} 1-\mathrm{O} 2^{\text {vii }}$ | 101.97 (16) |

Symmetry codes: (i) $x-1, y, z$; (ii) $-y+1 / 2, x-1, z$; (iii) $-y+1 / 2, x, z$; (iv) $-x,-y,-z+1$; (v) $-x,-y+1,-z+1$; (vi) $-x+1,-y,-z+1$; (vii) $-x+1,-y+1,-z+1$; (viii) $x, y, z+1$; (ix) $x, y+1, z$; (x) $-y+3 / 2, x, z$.

