

## 07.7-2 INTERCALATION OF NATURAL FLAKE GRAPHITES.

By I.A.S. Edwards, H. Marsh and N. Murdie, Northern Carbon Research Laboratories, School of Chemistry, The University of Newcastle upon Tyne NE1 7RU, England.

Intercalation compounds of pure single crystal graphites are well known and characterised (Ubbelohde, Chemistry and Industry, (1983) p. 636). Naturally occurring flake graphites also form intercalates but without the regularity or stoichiometry of the more ordered materials.

The intercalation abilities of a range of flake graphites with bromine and bisulphate have been studied. The effects of factors such as the distribution of mineral matter, flake thickness and fissuring have been examined and the stabilities of the intercalate assessed.

Semi-automatic image analysis has been used to measure the flake thickness, the degree of fissuring and the changes which take place on intercalation. Bromine intercalation and desorption has been measured and related to the crystal perfection of the flake graphite. Scanning electron microscopy, with elemental analysis, has been used to measure the distribution of mineral matter in the flakes, the distribution of crystallites and to assess the exfoliation of the graphite.

Results show that the flake thickness is a prime factor influencing the extent of intercalation in these materials, the optimum thickness for bromine intercalation being 25  $\mu\text{m}$ . Bromine uptake gives an indication of the perfection of stacking within the flakes of the graphite. Mineral matter within the flakes influences the thickness and cleavage properties.

## 07.8-1 INFRARED ABSORPTION OF LOCAL MODE CAUSED BY IMPURITY HYDROGEN IN SINGLE CRYSTAL SILICON By Gu Benyuan, Xu Zhenyi and Ge Peiwen, Institute of Physics, Academia Sinica, Beijing, China.

A series of characteristic infrared absorption lines within the range 1835 to 2220  $\text{cm}^{-1}$  have been observed in both single crystal silicon implanted with proton (Stein, J. Electronic Mat. (1975) 4, 159) and grown in a hydrogen atmosphere (Cui et al., to be published in Scientia Sinica). For single crystal silicon grown in a hydrogen atmosphere we suggest that there are four configurations of H in crystal.

A Si atom could be substituted by 1, 2, 3 or 4 H atoms which departed from the lattice site toward the nearest neighbour Si atoms and bonded with them. We designated them a, b, c and d, respectively. The vibration frequencies of local modes in these cases were calculated based on molecule model. The molecules consist of H atoms located at Si vacancy and the nearest neighbour Si. The relation between Si-Si force constants and valence force constants was presented and the Si-Si force constants were determined by the known Si valence force constants (Solbrig, Jr., J. Phys. Chem. Solids (1971) 32, 1761). In calculations we considered the nearest neighbour Si-H force constants and the influence of local lattice distortion and take all of them as variable parameters. Using group theory we can calculate infrared active modes. The comparison between calculated and experimental results is given in following table. Experimental data in the table are strong lines observed in all single crystal silicon grown in a hydrogen atmosphere. There are twenty-two H local modes which are infrared

active. Considering natural degeneracy there are only fourteen different frequencies. But ten of them can be classed in four groups, two of them are quasi-accidental twofold degeneracy, and the other two threefold. Consequently, there are only eight different frequencies. It is just equal to the numbers of strong lines observed. In the case of single crystal silicon implanted with proton, some kind of defects could arise from implanting. Therefore, their line numbers in infrared spectra are more than that grown in a hydrogen atmosphere.

Frequency ( $\text{cm}^{-1}$ )		Configu- ration	Point group	Representa- tion
Exper.	Cal.			
548	547	a	$C_{3v}$	E
634	658 658 658	b	$C_{2v}$	$A_1$ $B_1$ $B_2$
791	753 753 754	c	$C_{3v}$	$A_1$ E E
812	838	d	$T_d$	$T_2$
1949	1947	a	$C_{3v}$	$A_1$
1994	2038 2038	b	$C_{2v}$	$A_1$ $B_1$
2124	2125 2125	c	$C_{3v}$	$A_1$ E
2210	2208	d	$T_d$	$T_2$

## 07.8-2 INTERFERENCE ANALYSES OF MULTILAYERED THIN FILMS OF III-V COMPOUNDS BY MULTIPLE X-RAY DIFFRACTION. By C. Schiller, L.E.P. Limeil-Brevannes (France), W.J. Bartels, J. Hornstra and D.J.W. Lobeek, Philips Research Laboratories, Eindhoven (The Netherlands)

Crystal growers of thin films of III-V compound semiconductors need complete information about lattice parameters and thicknesses of epitaxial layers, uniformity of composition as a function of depth and about the shape of the concentration depth profile. This is essential for different applications, lasers or MIS structures and high-electron-mobility transistors.

X-ray double diffraction (Bartels and Nijman, J. Cryst. Growth, 44 (1978) 518) and X-ray multiple diffraction (Bartels, J. Vac. Sci. Technol. B, 1 (1983) 338) provide quick and non-destructive analyses of thin films in the range of 0.1 to 10  $\mu\text{m}$ . Anomalies in diffraction intensities are frequently observed when the stacking sequence increase in complexity, interference fringes commonly detected in monolayer thin films become more confusing with, in some cases, contrasts increasing up to 50 % as regards to a monolayer diffracted intensity.

The  $\text{Ga}_{1-x}\text{Al}_x\text{As}$  system was studied using MOCVD grown layers with values of x from 0 to 0.8, leading to a lattice parameter change from 0 to 2.1  $10^{-3}$  for  $\Delta a/a$  perpendicular to the growth plane.

The kinematical diffraction model has been applied to layer structures with total thicknesses in the range of 0.1 to 5  $\mu\text{m}$ : computing of diffracted intensities by Fourier summation of a function containing stacking sequence, number of layers, lattice parameter changes and concentration profiles has been realized.